

マグネシウム循環社会推進協議会 「Mg-Day in SENDAI Ⅲ」 (於：東北大学片平さくらホール)

講演4

「マグネシウム蓄電池(正極材料)の開発の経緯と現状迄の進展」

15:10~15:30

東北大学金属材料研究所構造制御機能材料学研究部門

教授 市坪 哲

2006 Mg蓄電池研究開始

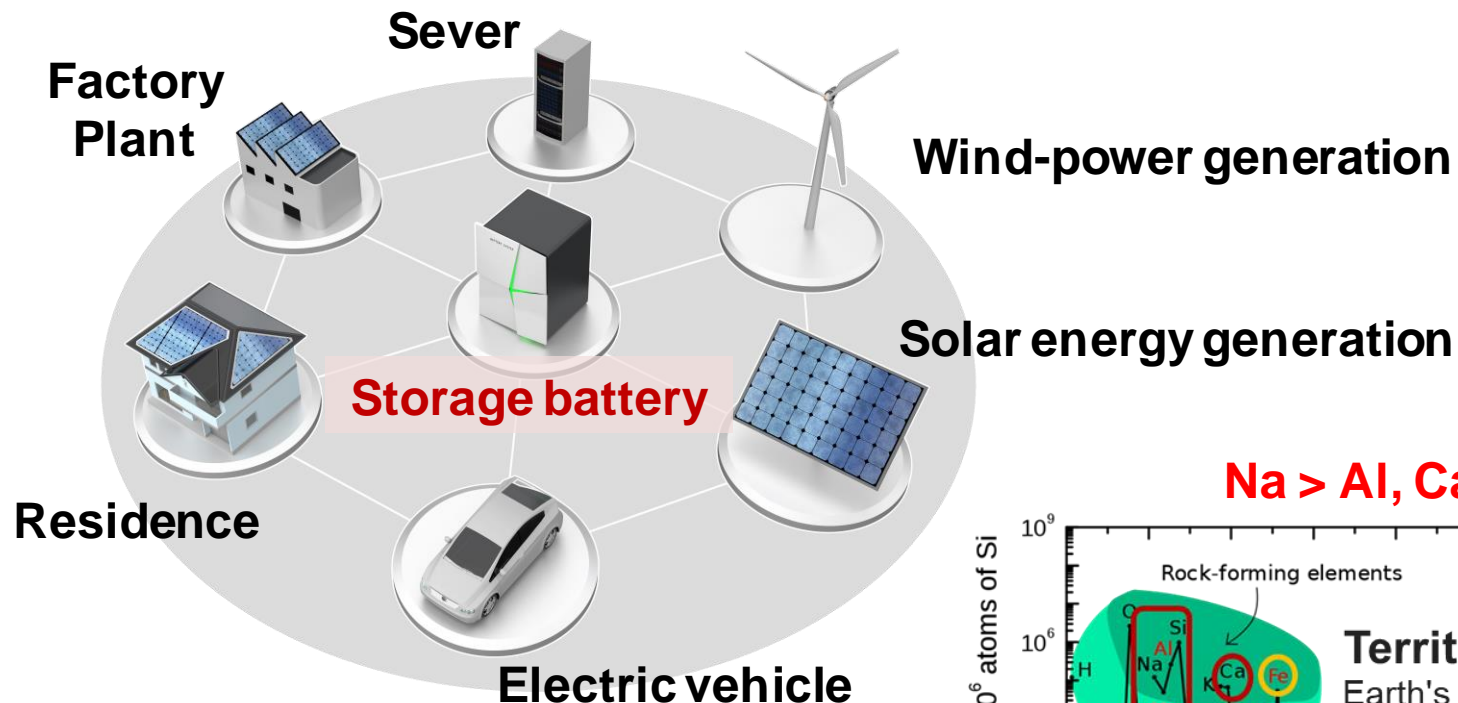
研究補助金:

ALCA(逢坂蓄電デバイス) 2011-2013

ALCA-SPRING移籍(金村TL)2013-2023 正極開発GLなど任務

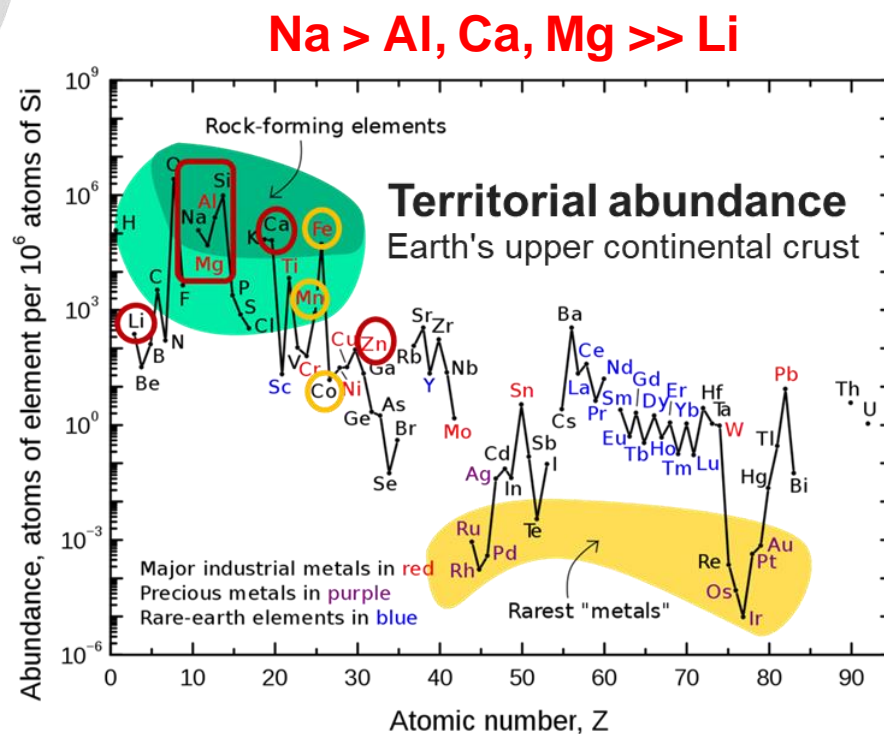
How to utilize the sustainable energy?

Smart-grid system for energy sustainability

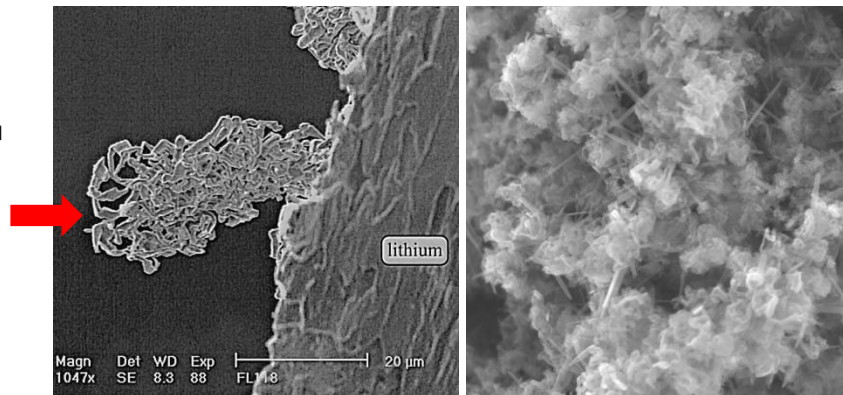
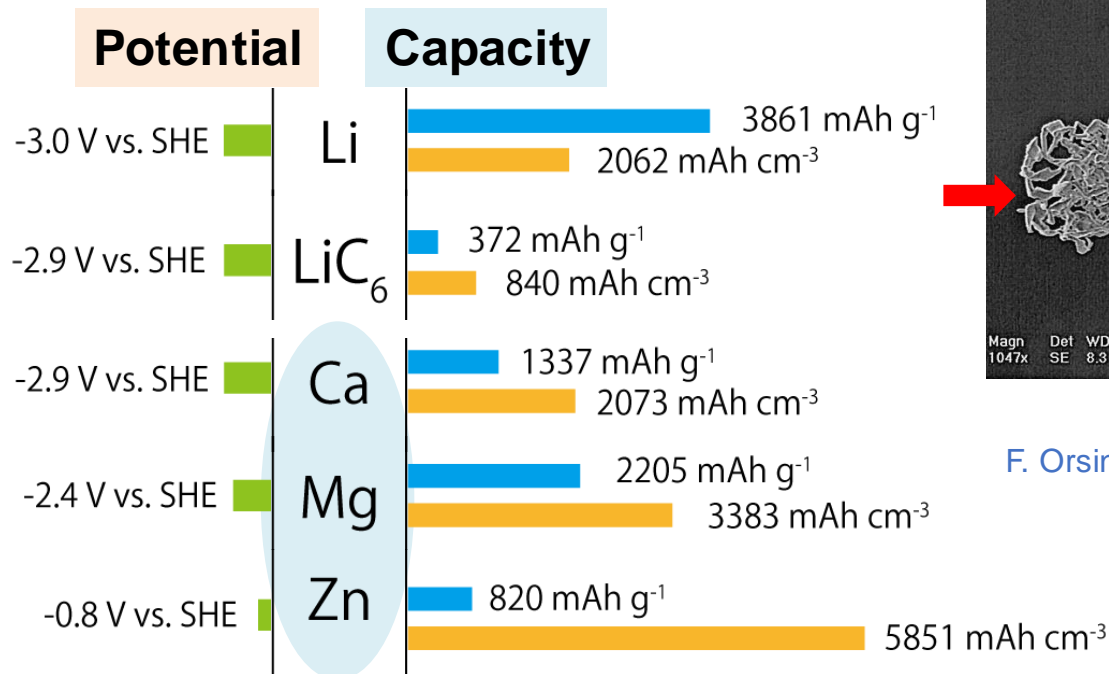


Necessary conditions for scaling-up

1. Resources must be **abundant!**
2. The most important thing is that rechargeable battery must be **safe!!**

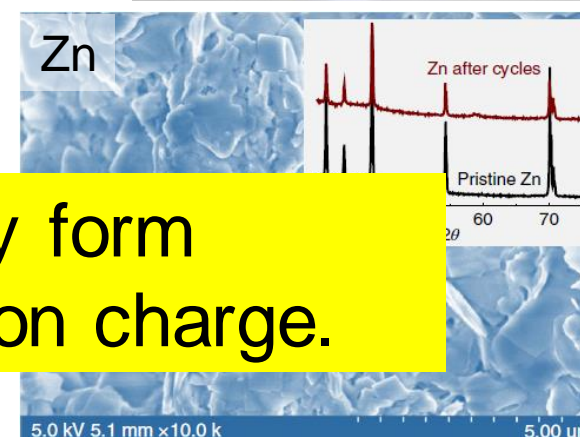
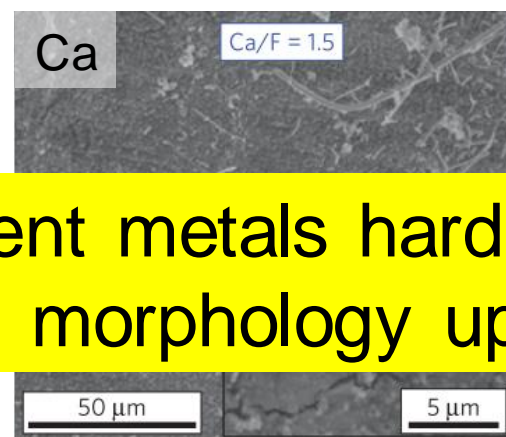
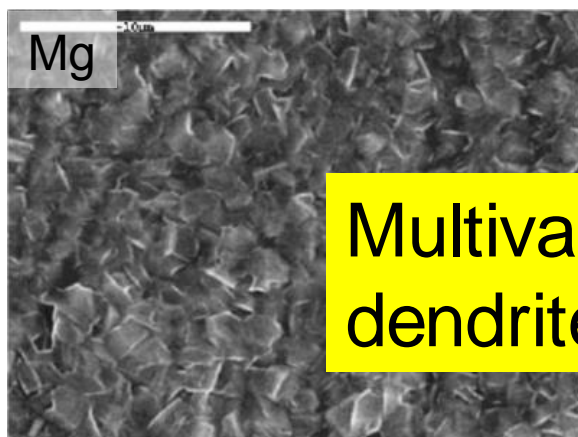
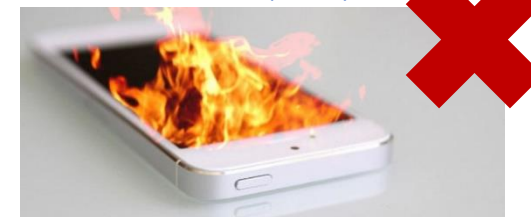


Various metal anodes and deposited morphology



Dangerous Li dendrites

F. Orsini et al, *J. Power Sources* **76**, 19 (1998).



Multivalent metals hardly form dendrite morphology upon charge.

O. Mizrahi et al, *J. Electrochem. Soc.*, **155**, A103 (2008).

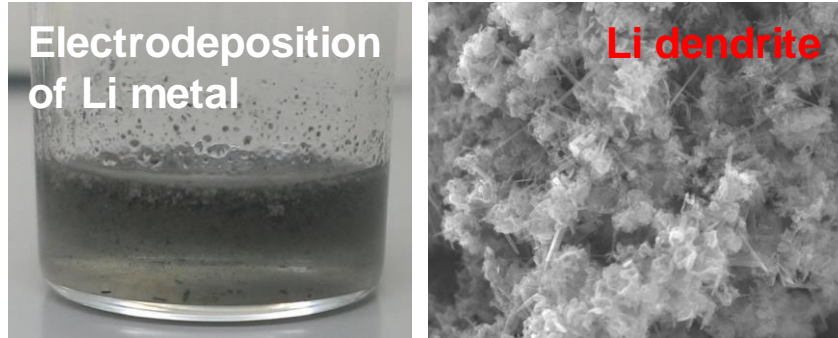
A. Ponrouch et al, *Nat. Mater.* **15**, 169 (2015).

F. Wang et al, *Nat. Mater.* **17**, 543 (2018).

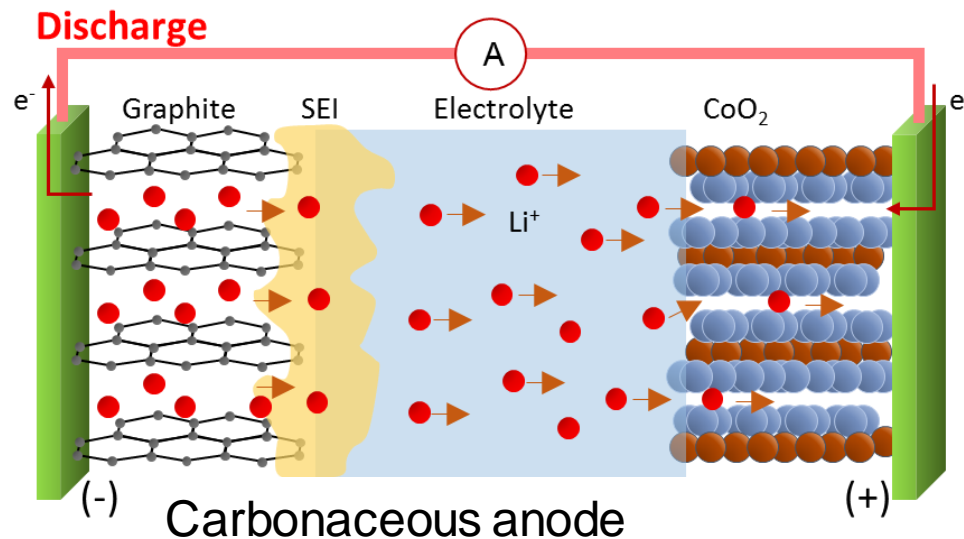
Why magnesium rechargeable battery (MRB) ?

Lithium ion battery (LIB)

Li metal cannot be used as anode material due to a fatal problem, i.e., **dendrite** electrodeposition of Li metal.

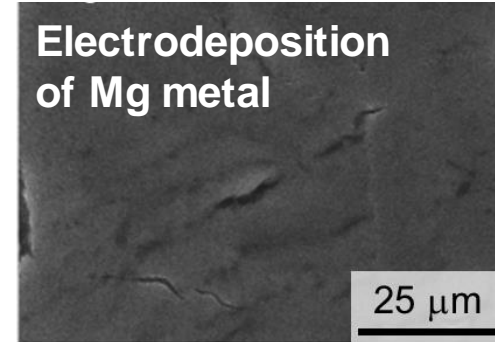


LiC_6 : **72g-carbon** per 1 mol-electron.
(3.7V-140Ah/kg | EC/DMC | Graphite)
→ 370 Wh/kg

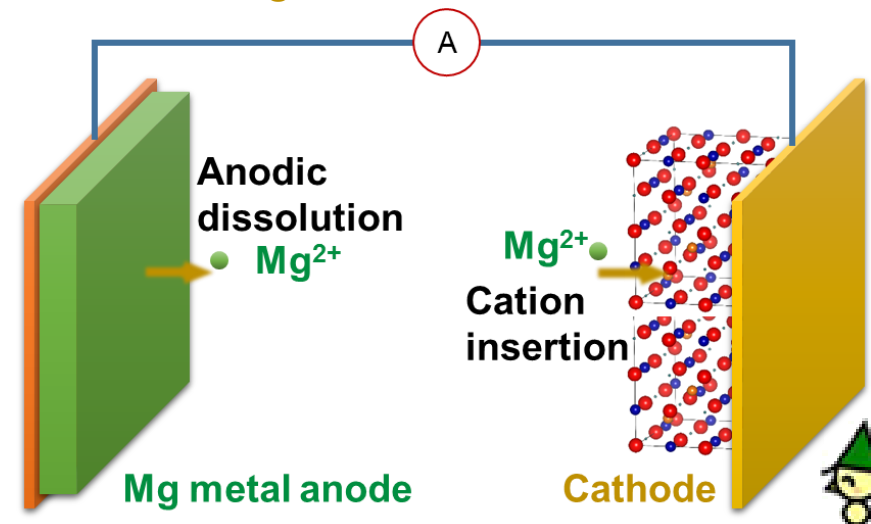


Magnesium rechargeable battery (MRB)

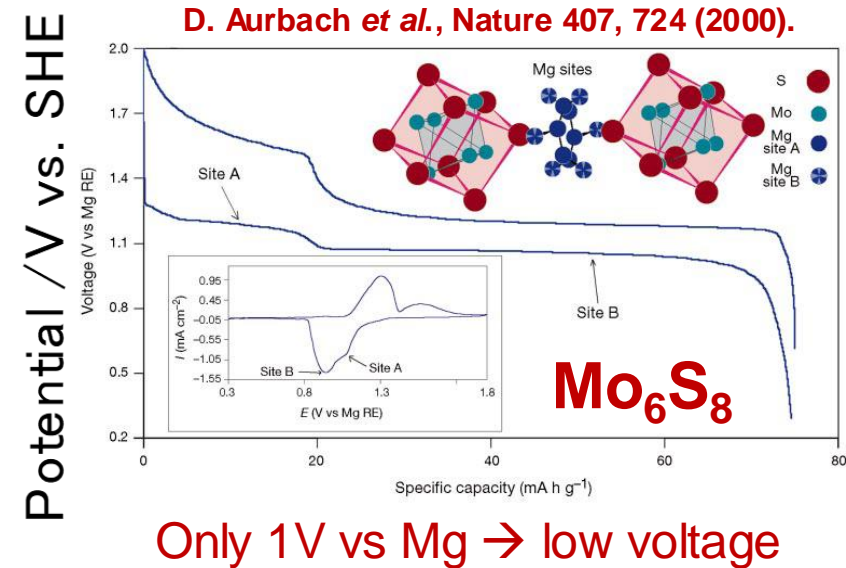
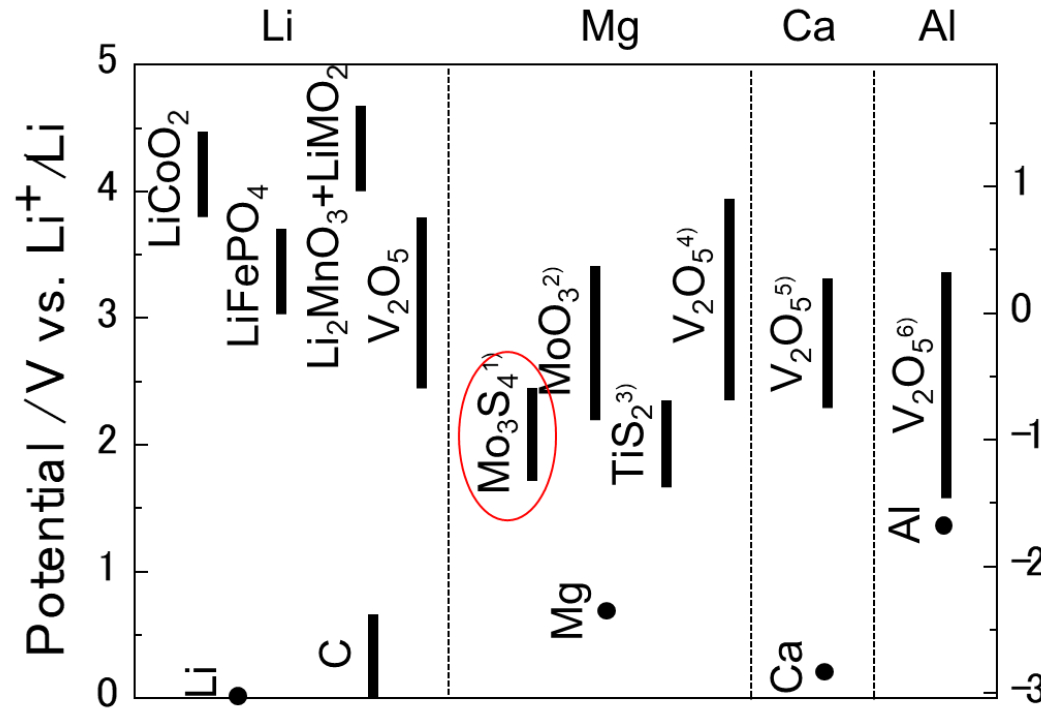
Mg metal can be used as anode material, because Mg metal is **electrodeposited smoothly** on current collector surface.



Only **12g-Mg** for 1 mol-electron
(2.8V-200Ah/kg | Electrolyte | Mg-metal)
→ 510 Wh/kg



§ 2. Cathode material candidates for rechargeable batteries



Many cathode candidates for LIB,
 but few materials for multivalent cations!

1) D. Aurbach *et al.*, Nature **407**, 724 (2000).

2) M. E. Spahr *et al.*, J. Power Sources **54**, 346 (1995).

3) Z. Tao *et al.*, Chem. Commun. **40**, 2080 (2004).

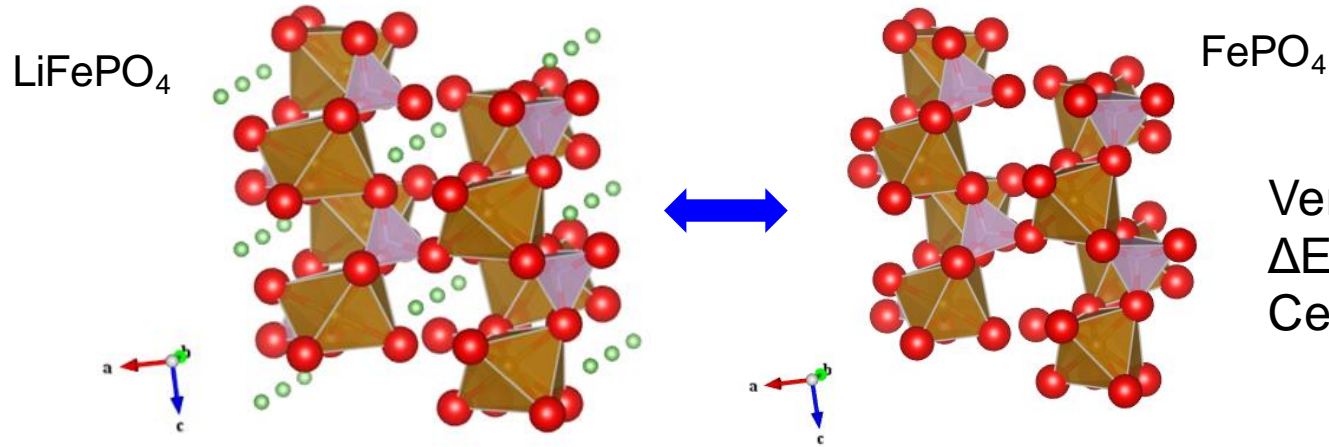
4) P. Novak *et al.*, J. Electrochem. Soc. **140**, 140 (1993).

5) M. Hayashi *et al.*, J. Power Sources **119**, 617 (2003).

6) N. Jayaprakash *et al.*, Chem. Commun. **47**, 12610 (2012).

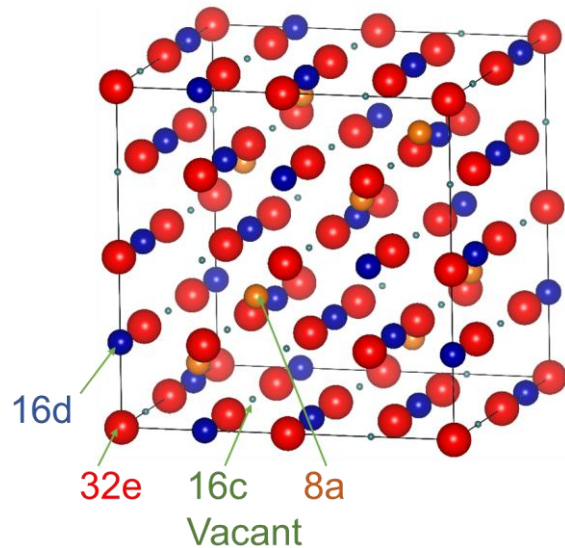
§ 2. Activation energy movement: Divalent ions vs Li ion

Olivine : LiFePO_4



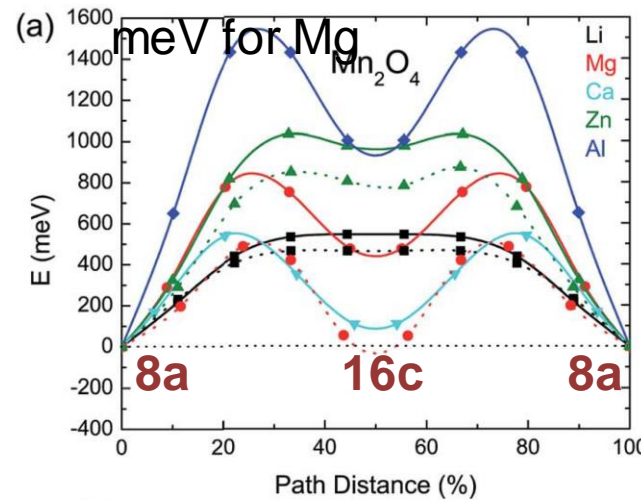
Very fast transition
 $\Delta E(\text{diffusion}) \sim 200 \text{ meV}$
 Ceder et al.

Spinel : AB_2O_4

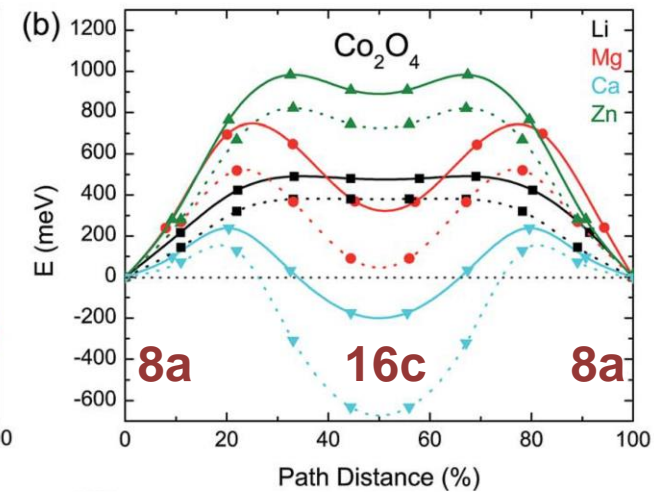


ab-initio calculation by Ceder et al.

$\Delta E \sim 400 \text{ meV}$ for Li

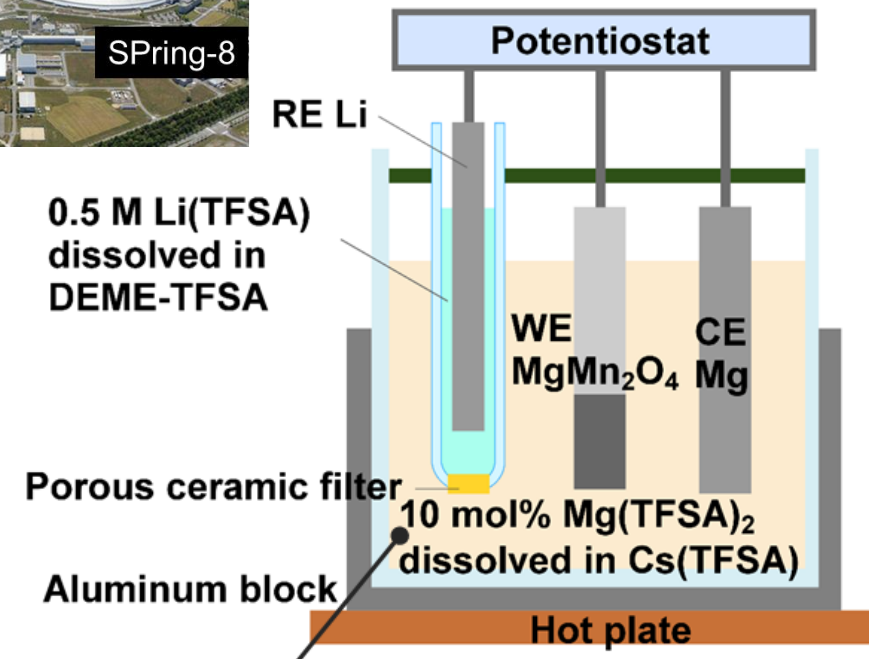
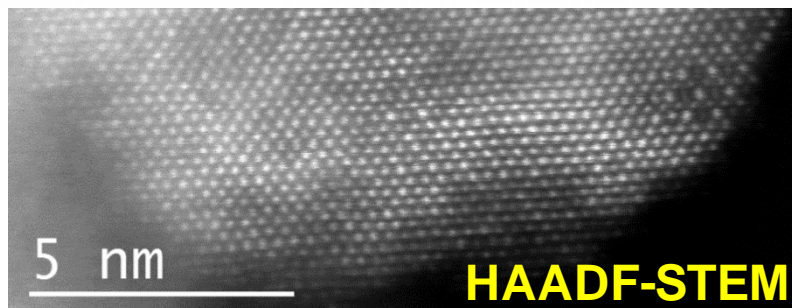
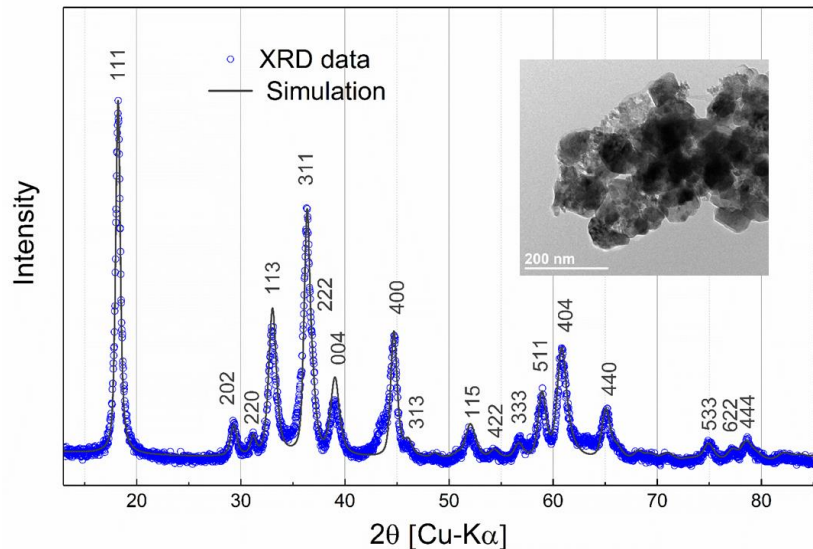


$\Delta E \sim 600-800$



Sample preparation and electrochemical method

1. Electrochemical measurement at moderate temperatures
→ using Mg/Cs-TFSA ionic liquid ($\sim 150^\circ\text{C}$)
2. Inverse co-precipitation method
→ Very fine (primary) particles of about 5-10 nm
3. Sample characterization
→ Synchrotron XRD, XAFS, XPS, STEM, ICP



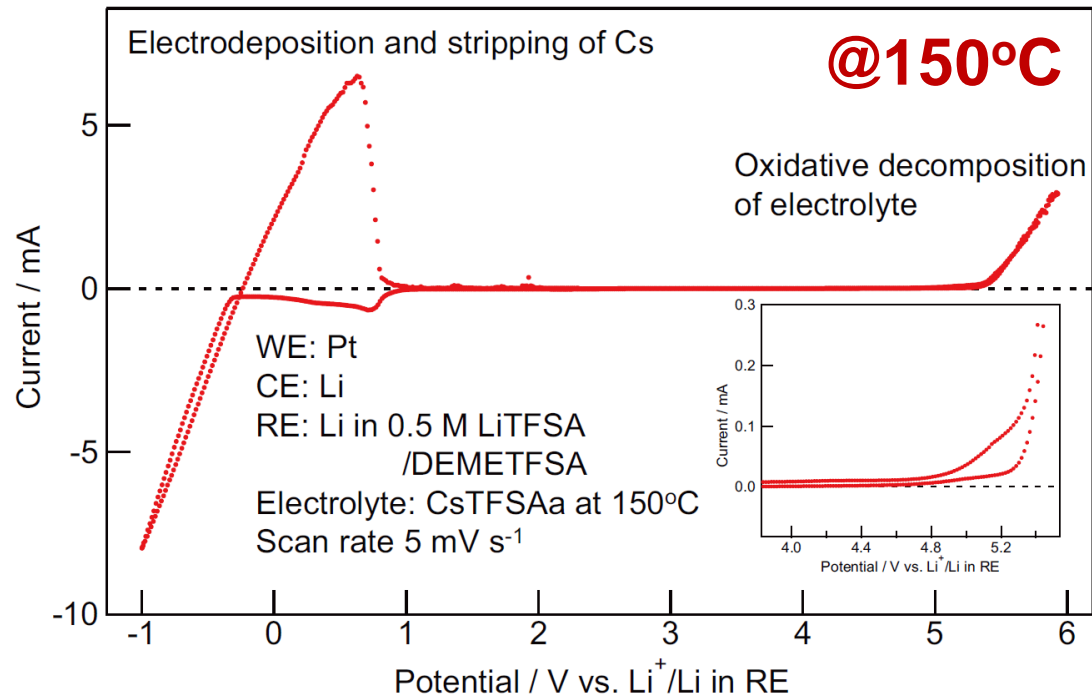
**Heated up to 150°C
to enhance Mg diffusion**

Moderate-temperature ionic liquid

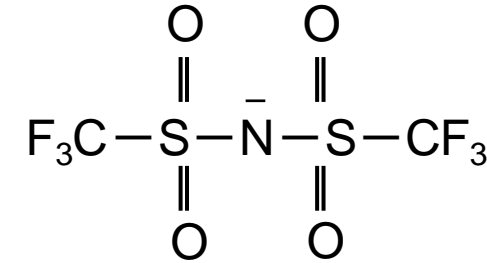
Cs-TFSA based ionic liquid
by Prof. Hagiwara et al.

Molten Salts Chemistry and Technology, M. Gaune-Escard and
G. M. Haarberg eds., John Wiley & Sons, Ltd. Chapt. 5.4 (2014).

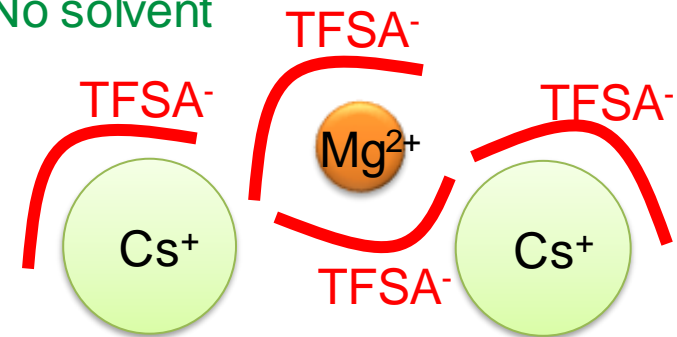
Melting temperature of Cs-TFSA: 120°C



TFSA⁻ anion



No solvent



Advantageous points:

1. Possible to operate at moderate temperatures (120-200 °C)
2. Wide electrochemical window:
High oxidation limit (~5V vs Li)



Estimating Redox Potential of Mg in (Mg10/Cs90)-TFSA liquid

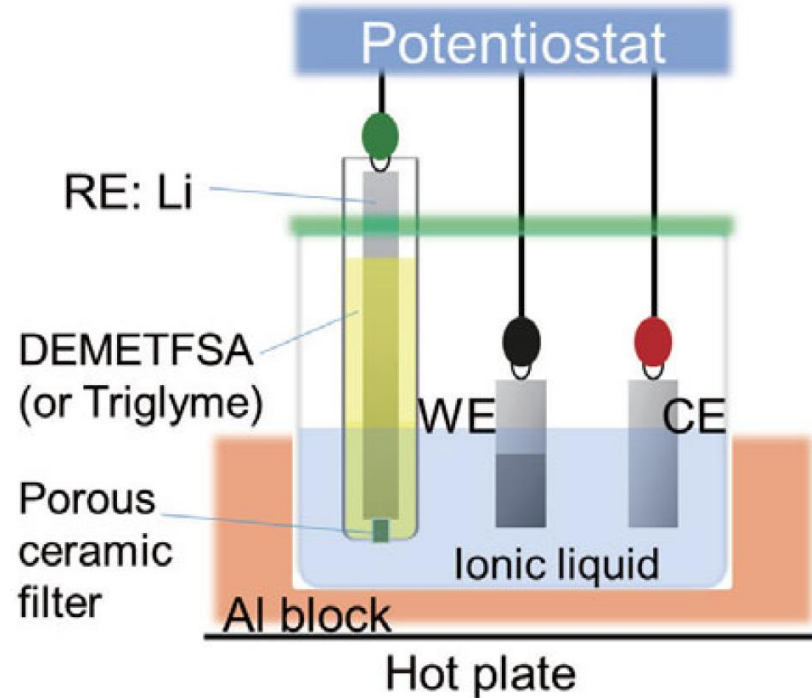
Mg(TFSA)₂ / CsTFSA = 10 / 90

Ar atmosphere

Dew point: -70 ~ -80 °C

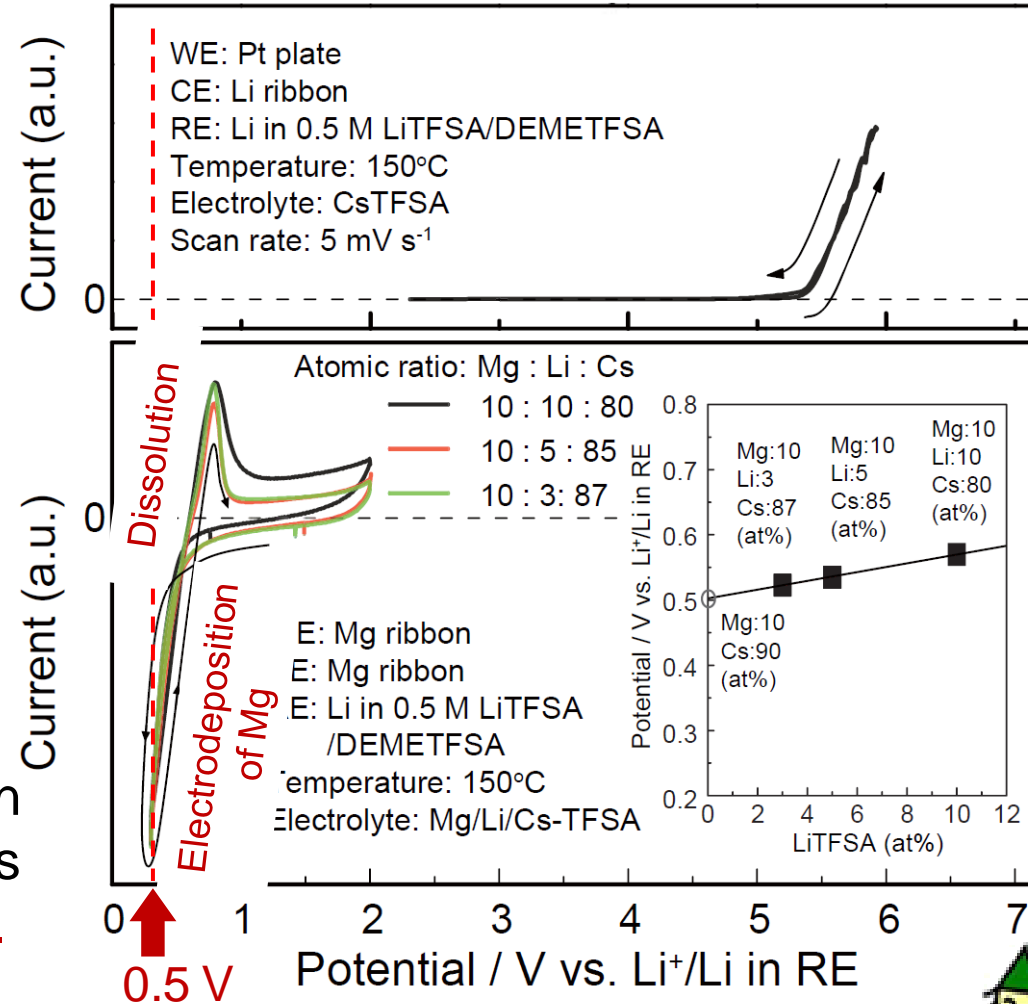
Reference electrode (RE): Li in 0.5M LiTFSA / DEME-TFSA solution

Counter electrode (CE): Mg ribbon



Redox potential conversion rule

The redox potential of Mg/Mg²⁺ in (Mg10/Cs90)-TFSA ionic liquid is estimated to be **0.5 V vs Li (in RE)**.

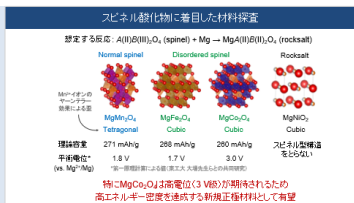




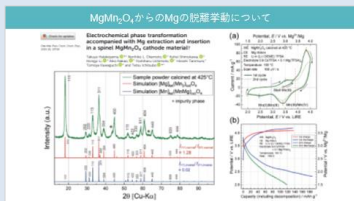
Mg蓄電池用正極材料の新しい設計指針の確立

東北大金研 市坪 哲, 下川 航平, 富山 拓也, 岡本 範彦

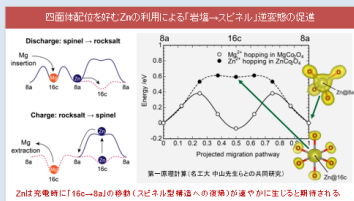
1. スピネル酸化物に着目した材料探査:スピネル-岩塩相転移の発見



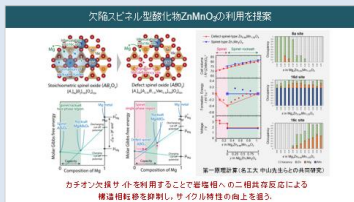
2. スピネル酸化物 MgMn₂O₄からのMg脱離の実行可能性



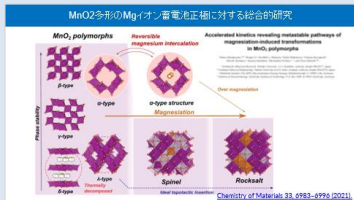
3. Zn基スピネル酸化物:可逆性の向上



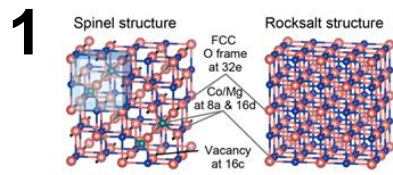
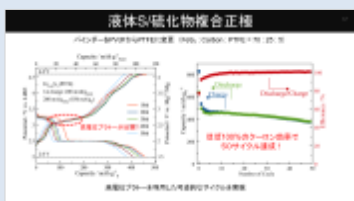
4. 欠陥スピネル酸化物 ZnMnO₃の高サイクル特性



5. MnO₂多形に対するMg挿入相変態の研究



6. TMS₂からのTM脱離によるS正極の開発

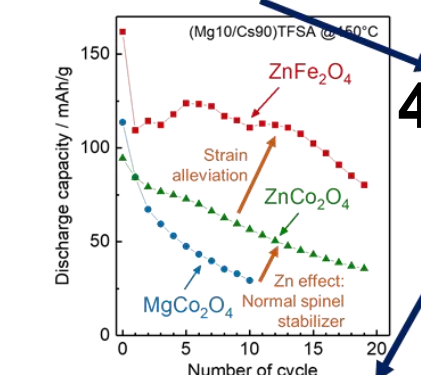


酸化物
高電位低容量低レート

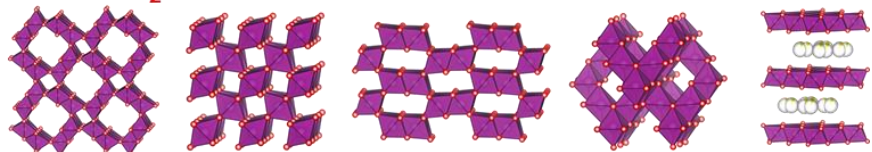
VS

硫化物
低電位高容量高レート

Spinel → Rocksalt



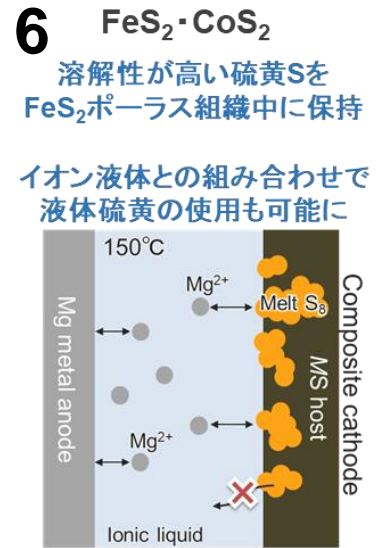
5 XO₂ 多形構造の活用
X = Cr, V, Ti, Mn
MnO₂
A(II)B(IV)O₃
[A_{1.0}]_{8a}[A_{0.33}B_{1.33}□_{0.34}]_{16d}O₄



XO₂ + AO
MnO₂へのAO添加によりさらなる多形構造へ展開
山中G, 小林Gらと共同検討

16d欠陥スピネル・スピネル間相転移
Topotacticな領域の活用で可逆性がさらに向上

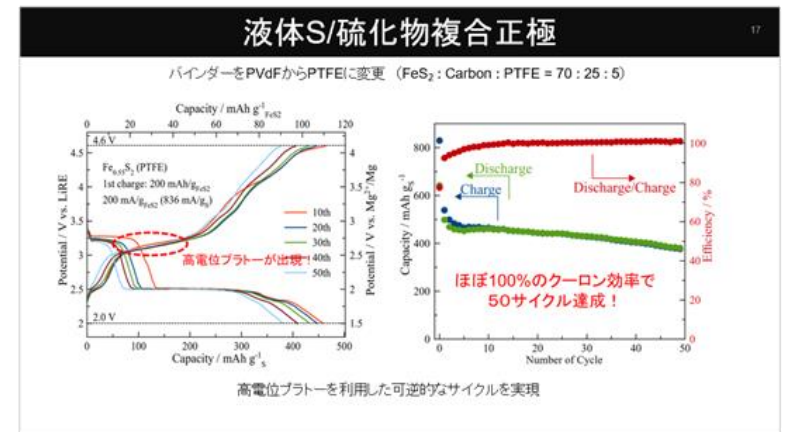
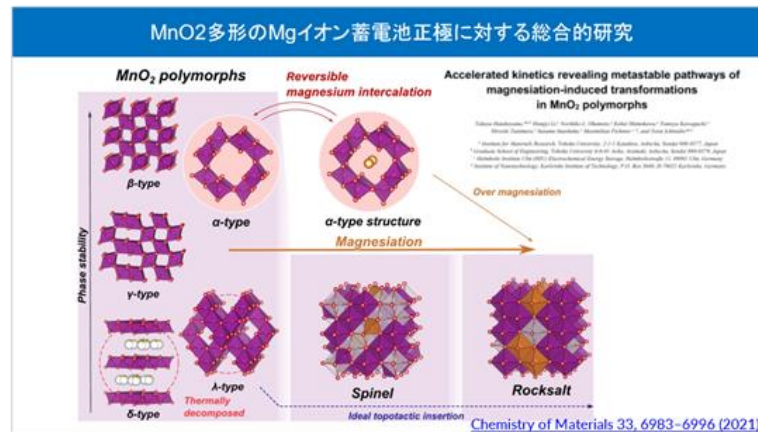
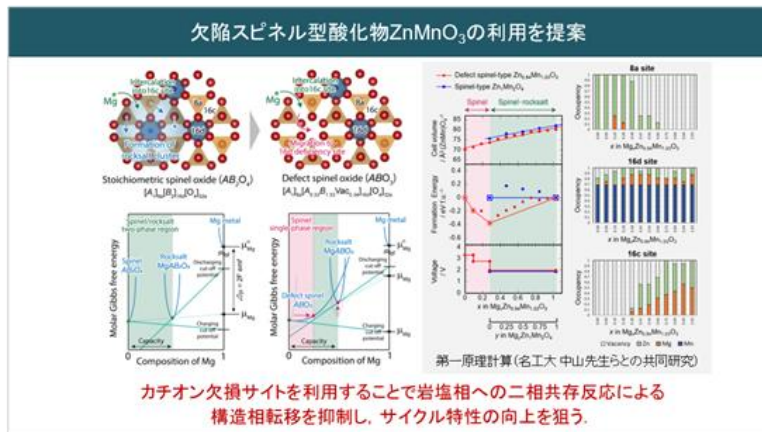
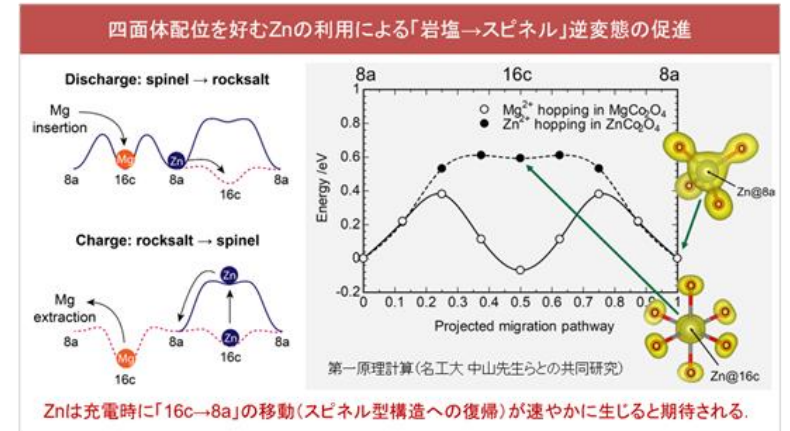
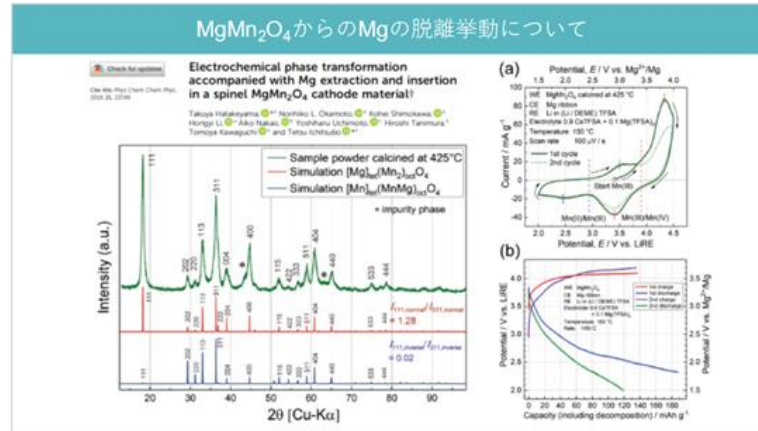
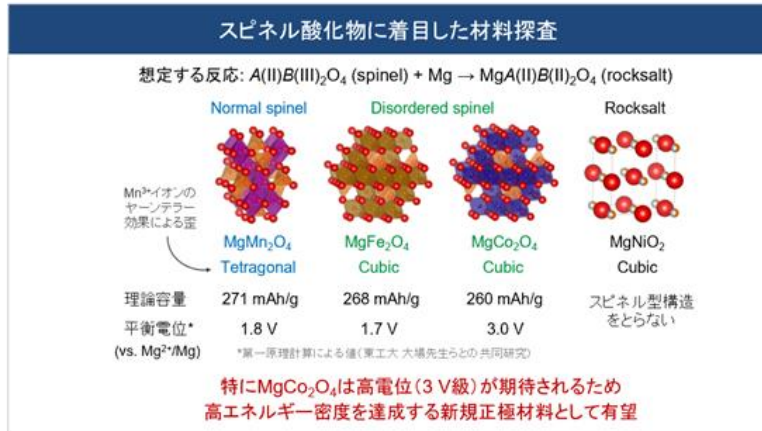
新しい化合物候補群を生み出す
演繹的な考え方の創出
↓
室温可動正極・設計指針の確立



今後の進展へ向けて

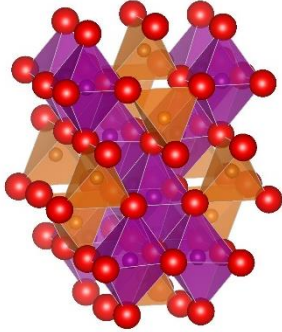
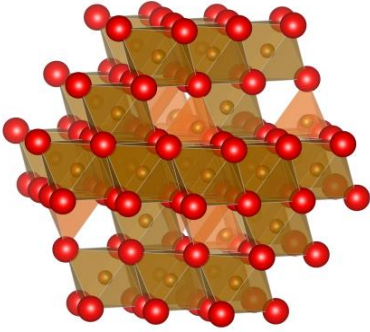
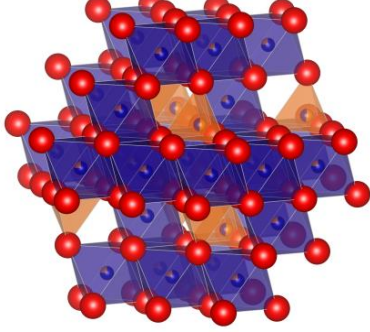
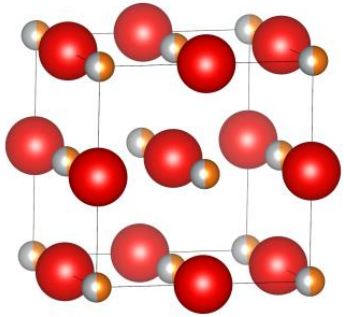
- 容量の増大化 100 → 200 or 300 Ah/kg
MnO₂ベースで開発促進
- Mg脱挿入の低温度化(できれば室温)
- HFIP系電解液での可逆的な脱挿入の実現
- デュアルカチオンの検討

説明用の詳細スライド



スピネル酸化物に着目した材料探査

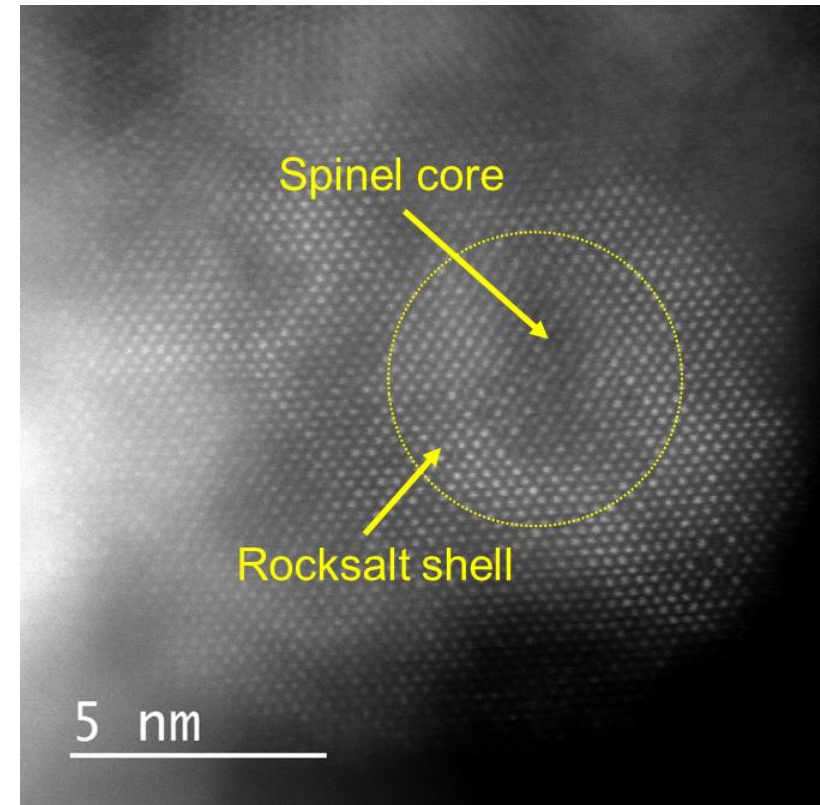
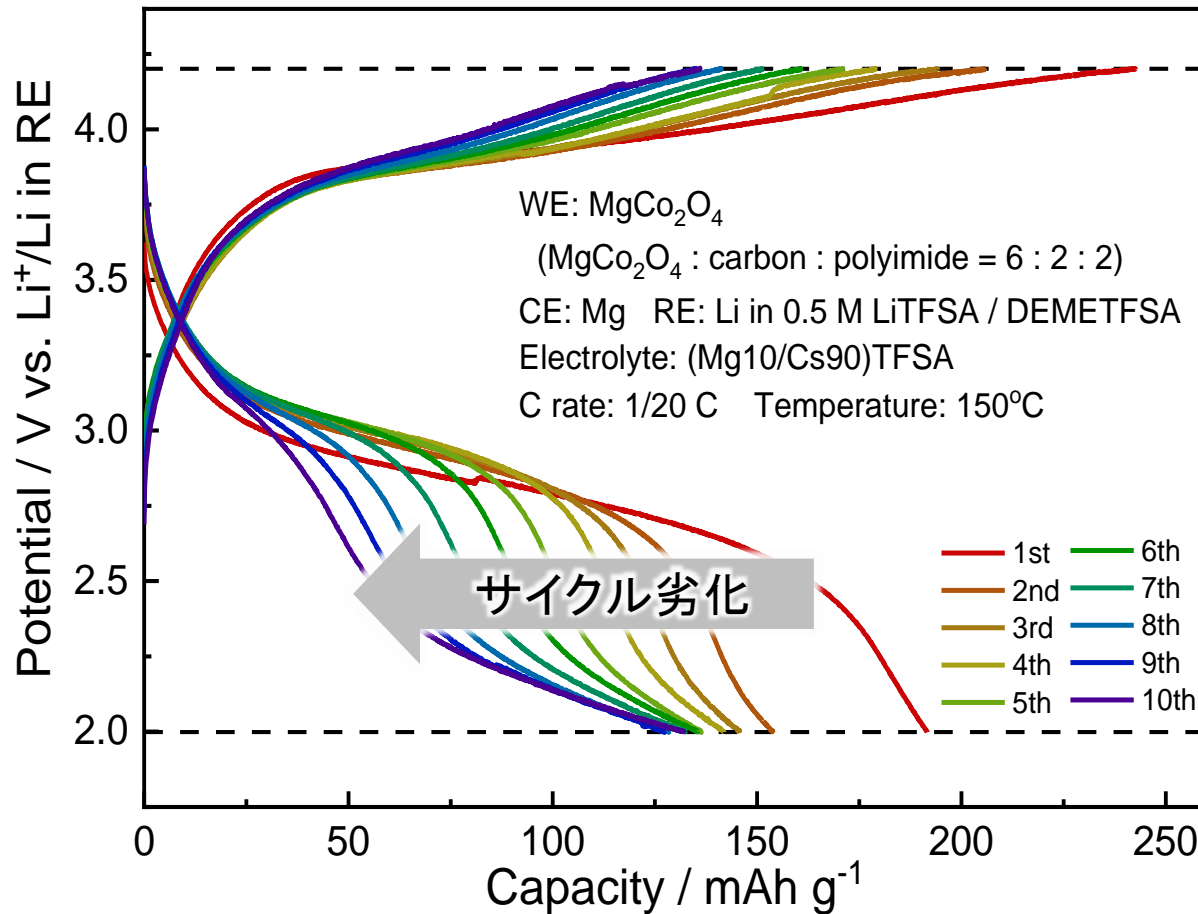
想定する反応: $A(II)B(III)_2O_4$ (spinel) + Mg \rightarrow MgA(II)B(II) $_2O_4$ (rocksalt)

	Normal spinel	Disordered spinel		Rocksalt
				
	Mn ³⁺ イオンの ヤーンテラー 効果による歪			
	MgMn₂O₄ Tetragonal	MgFe₂O₄ Cubic	MgCo₂O₄ Cubic	MgNiO₂ Cubic
理論容量	271 mAh/g	268 mAh/g	260 mAh/g	スピネル型構造 をとらない
平衡電位* (vs. Mg ²⁺ /Mg)	1.8 V	1.7 V	3.0 V	
	*第一原理計算による値(東工大 大場先生らとの共同研究)			

特にMgCo₂O₄は高電位(3 V級)が期待されるため
高エネルギー密度を達成する新規正極材料として有望

スピネル酸化物に着目した材料探査

スピネル相と岩塩相の二相共存反応(酸素格子は整合)により充放電が進行



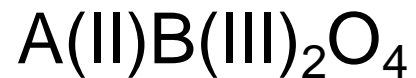
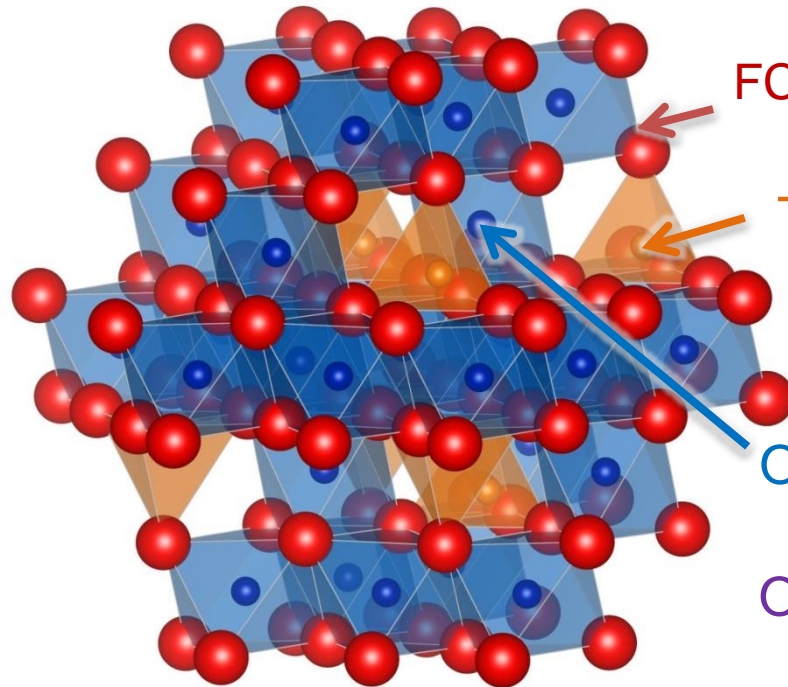
放電後の粒子のSTEM観察結果

サイクル劣化抑制に向けたスピネル型酸化物の新たな設計指針が必要

Structural similarity between Spinel and Rocksalt

Space group #227

Spinel



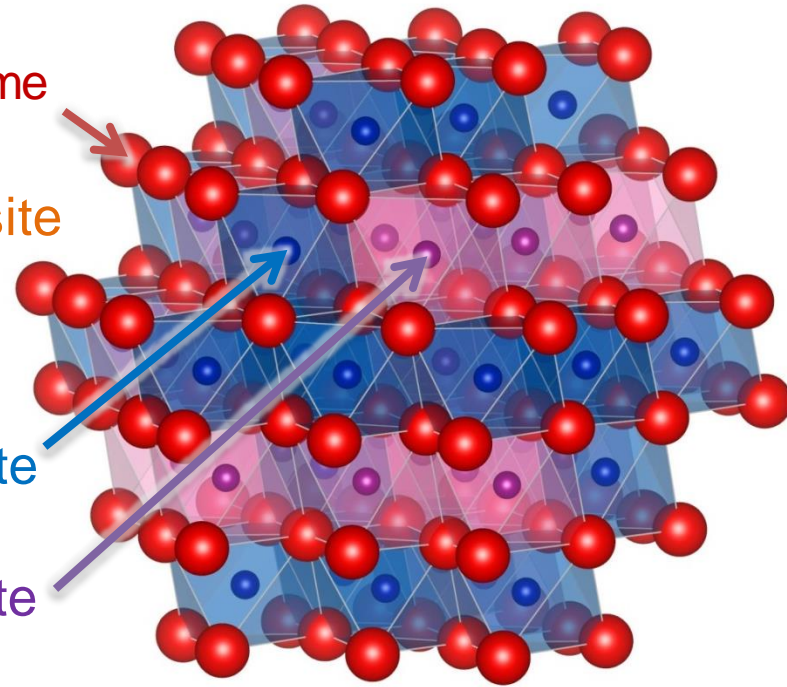
FCC oxygen flame
(32e site)

Tetrahedral site
(8a site)

Octahedral site
(16d site)

Octahedral site
(16c site)

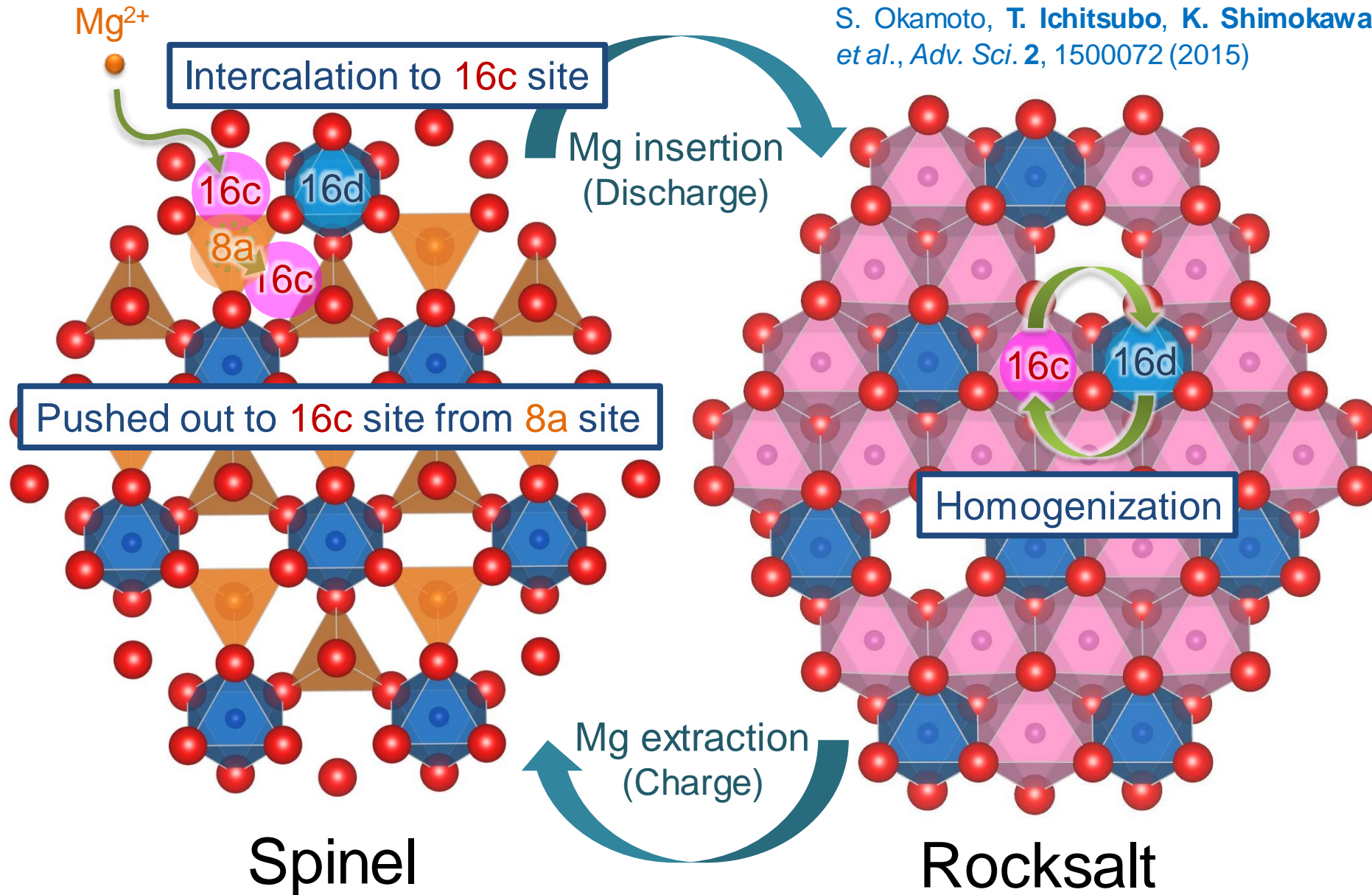
Rocksalt



Only difference is the configuration of 8a and 16c

“Intercalation and push-out” mechanism

S. Okamoto, T. Ichitsubo, K. Shimokawa
et al., *Adv. Sci.* **2**, 1500072 (2015)





Intercalation and Push-Out Process with Spinel-to-Rocksalt Transition on Mg Insertion into Spinel Oxides in Magnesium Batteries

Shinya Okamoto, Tetsu Ichitsubo, Tomoya Kawaguchi, Yu Kumagai, Fumiyasu Oba, Shunsuke Yagi, Kohei Shimokawa, Natsumi Goto, Takayuki Doi, and Eiichiro Matsubara*

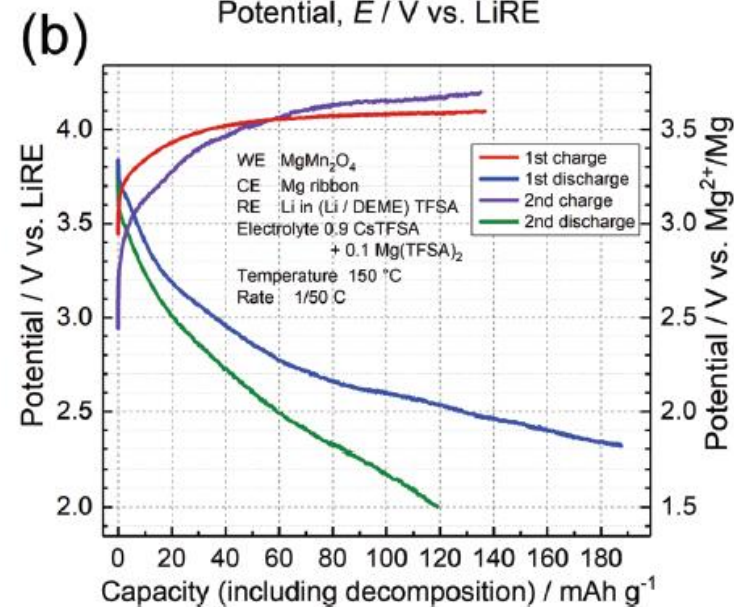
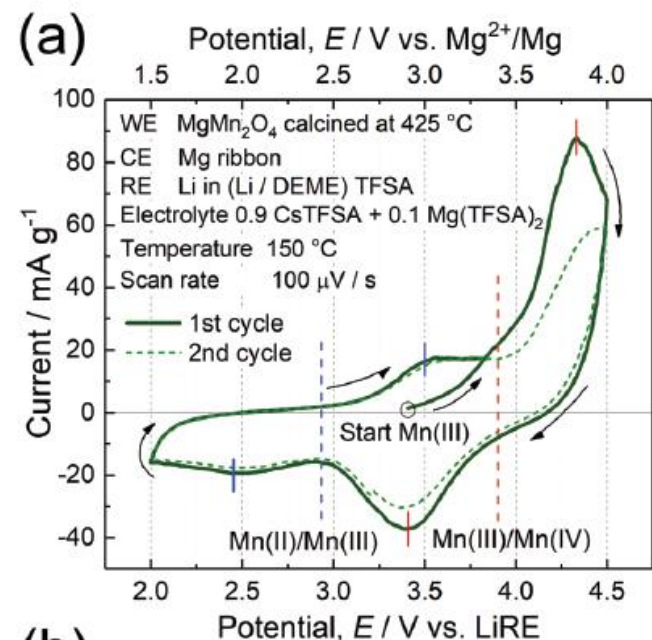
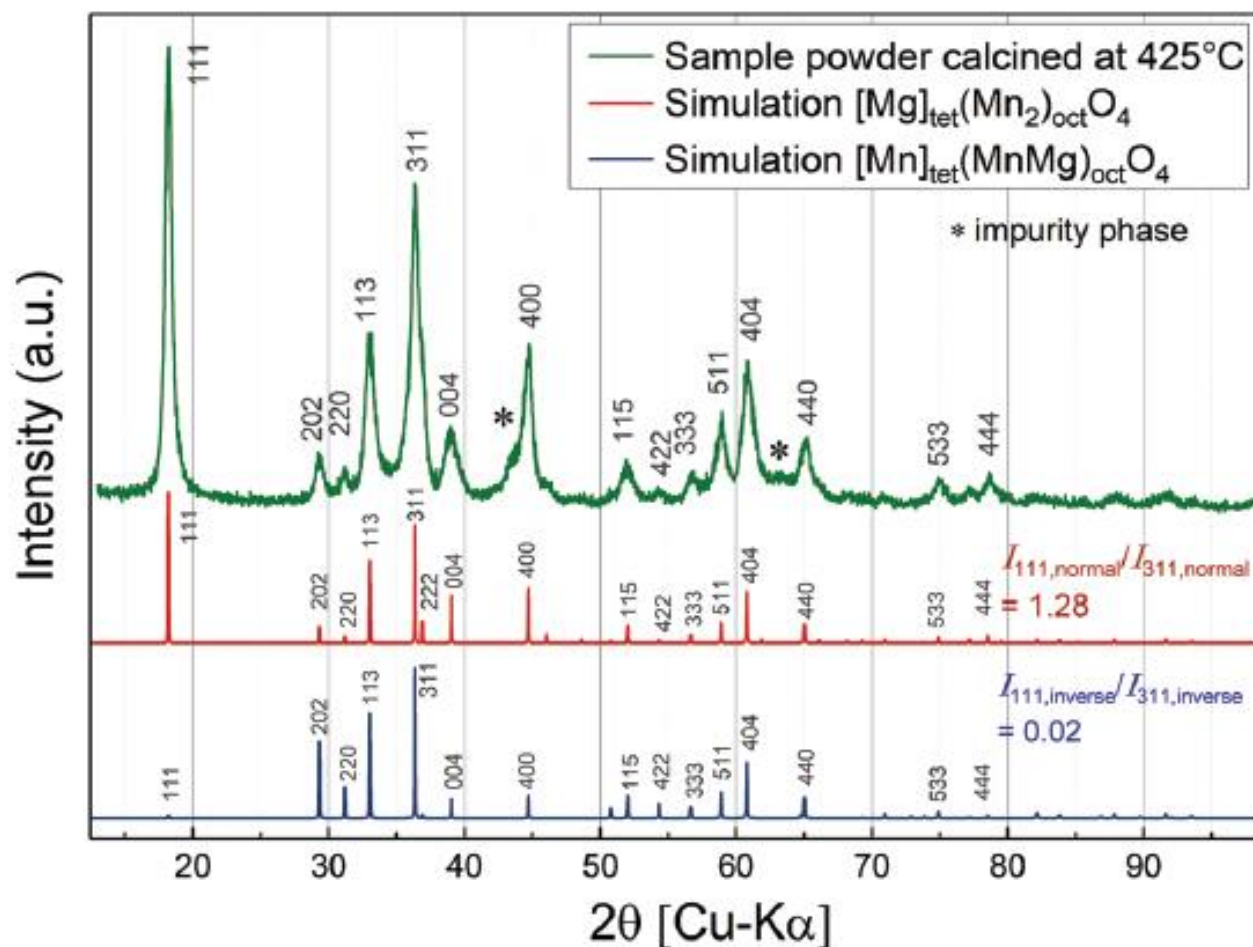
MgMn₂O₄からのMgの脱離挙動について



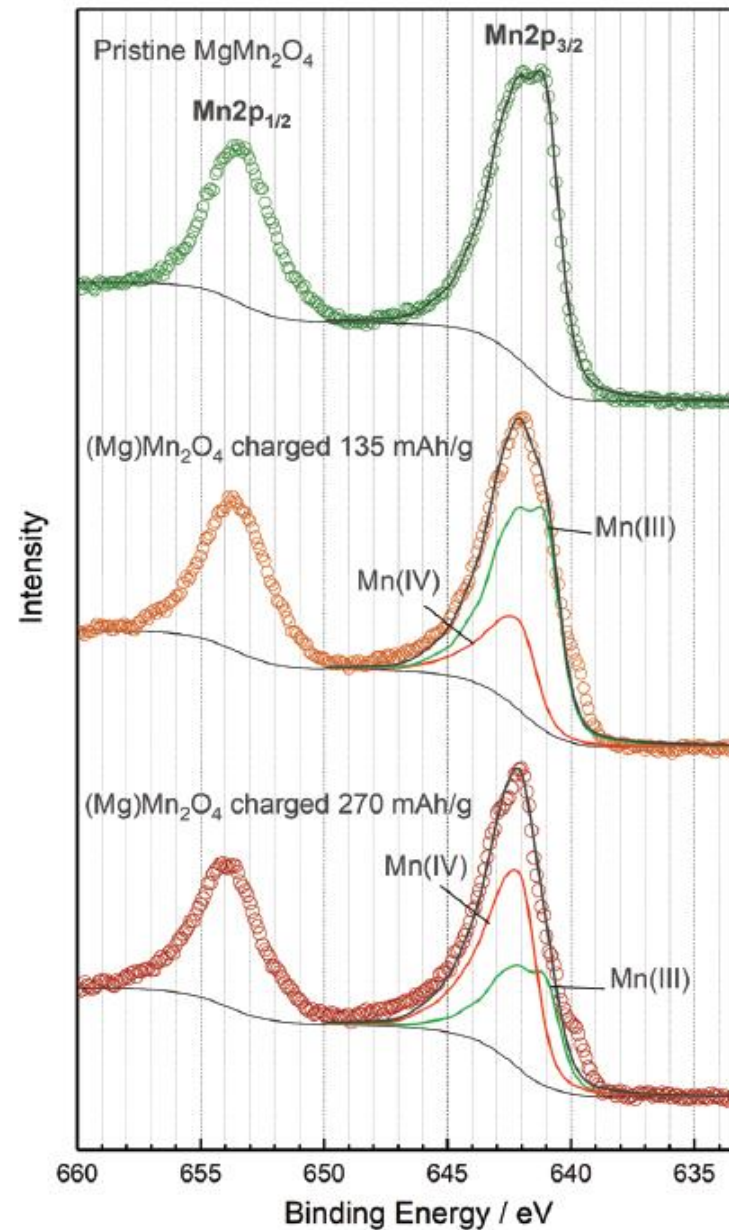
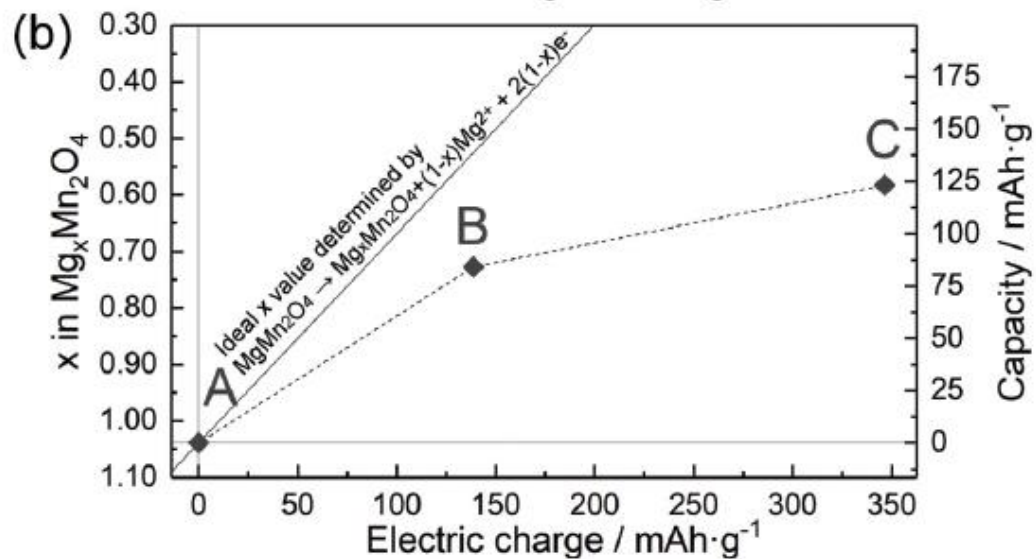
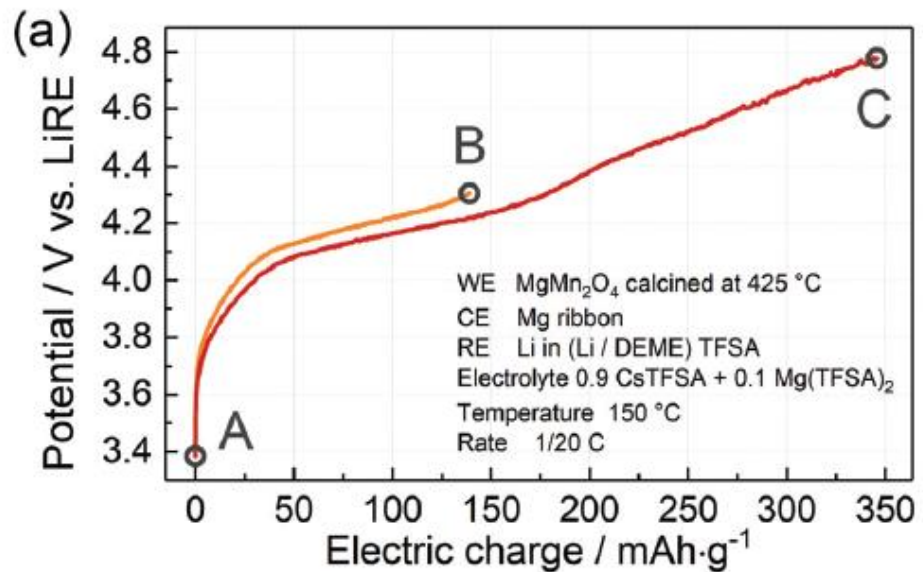
Cite this: *Phys. Chem. Chem. Phys.*, 2019, 21, 23749

Electrochemical phase transformation accompanied with Mg extraction and insertion in a spinel MgMn₂O₄ cathode material†

Takuya Hatakeyama, ^{ib}*^a Norihiko L. Okamoto, ^{ib}^a Kohei Shimokawa, ^{ib}^a Hongyi Li, ^{ib}^a Aiko Nakao, ^{ib}^b Yoshiharu Uchimoto, ^{ib}^c Hiroshi Tanimura,^a Tomoya Kawaguchi ^{ib}^a and Tetsu Ichitsuho ^{ib}*^a

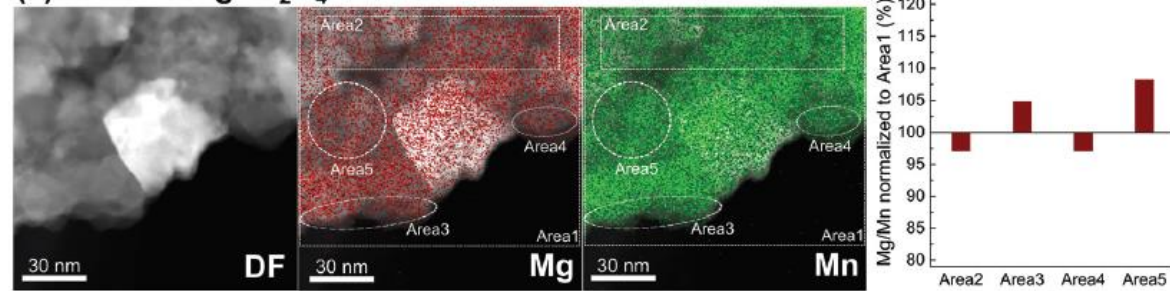


MgMn₂O₄からのMgの脱離挙動について

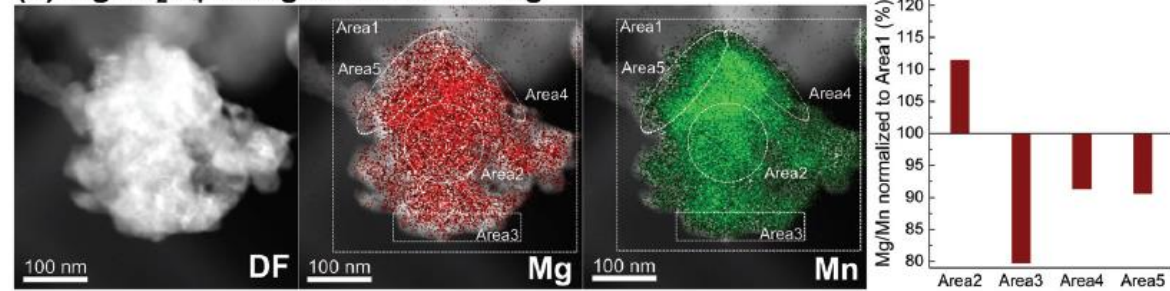


MgMn₂O₄からのMgの脱離挙動について

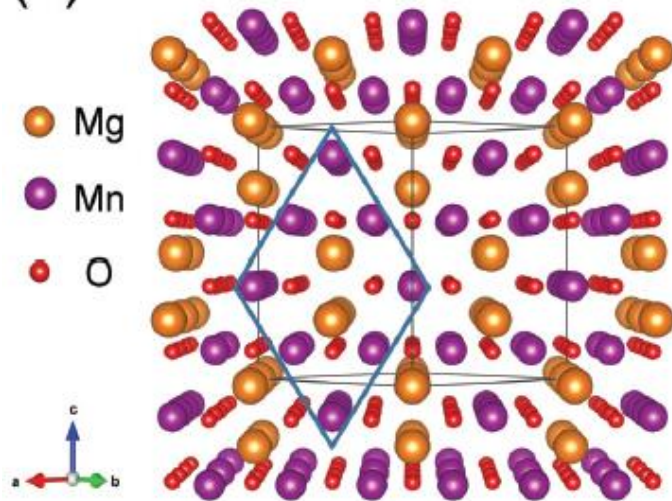
(a) Pristine MgMn₂O₄



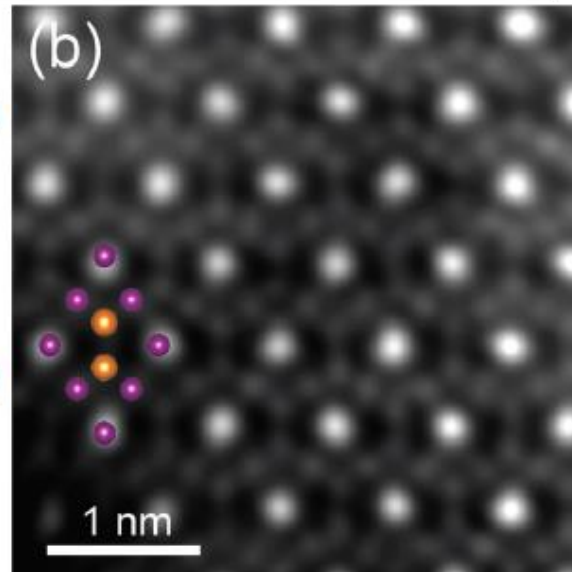
(b) MgMn₂O₄ charged to 270 mAh/g



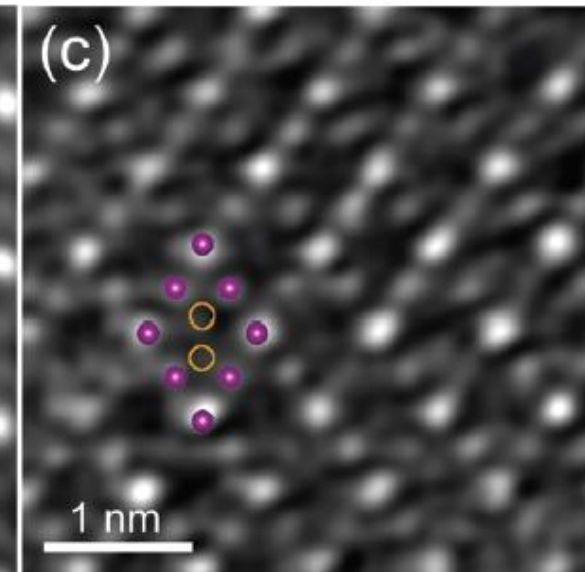
(a)



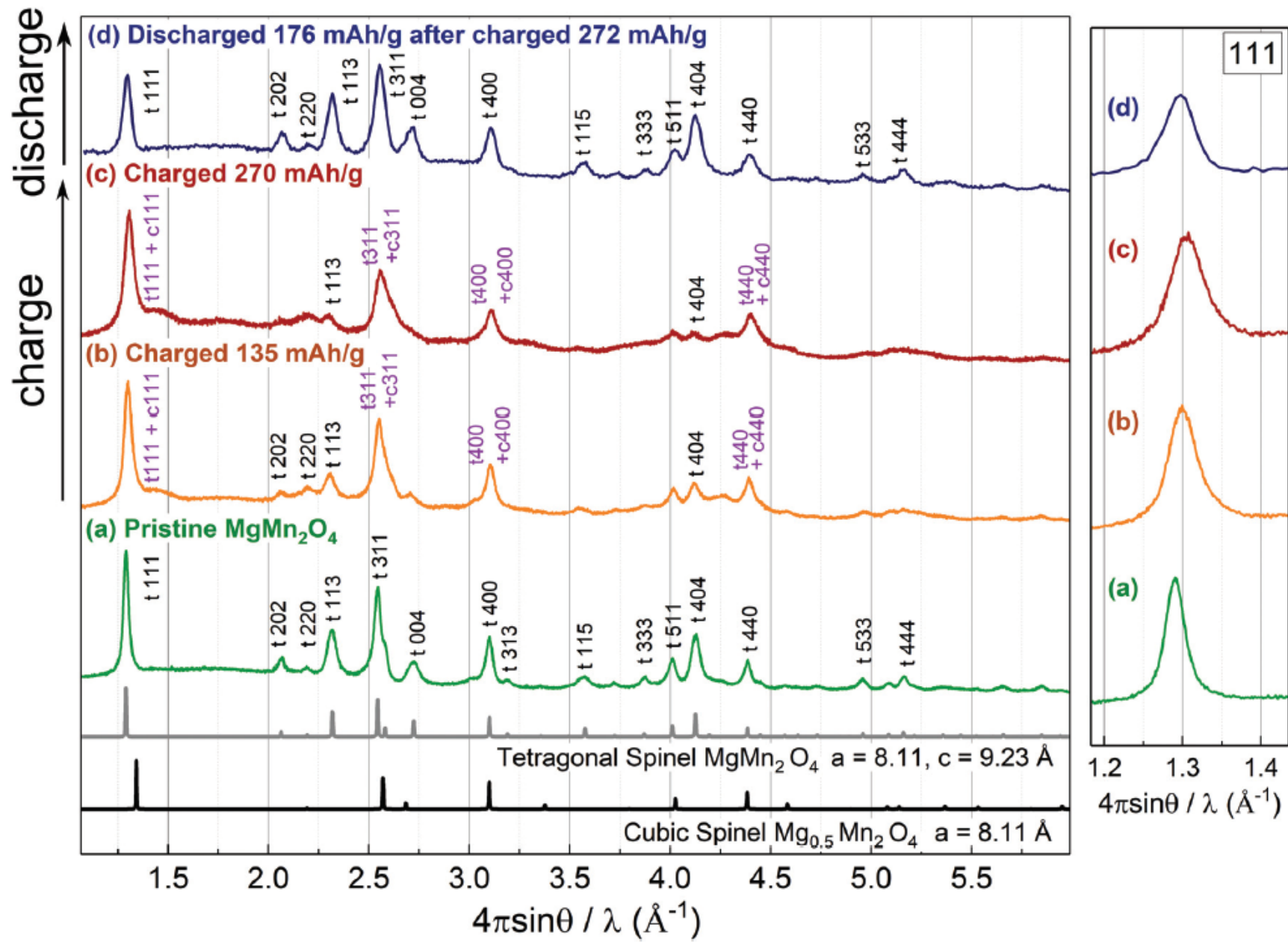
(b)



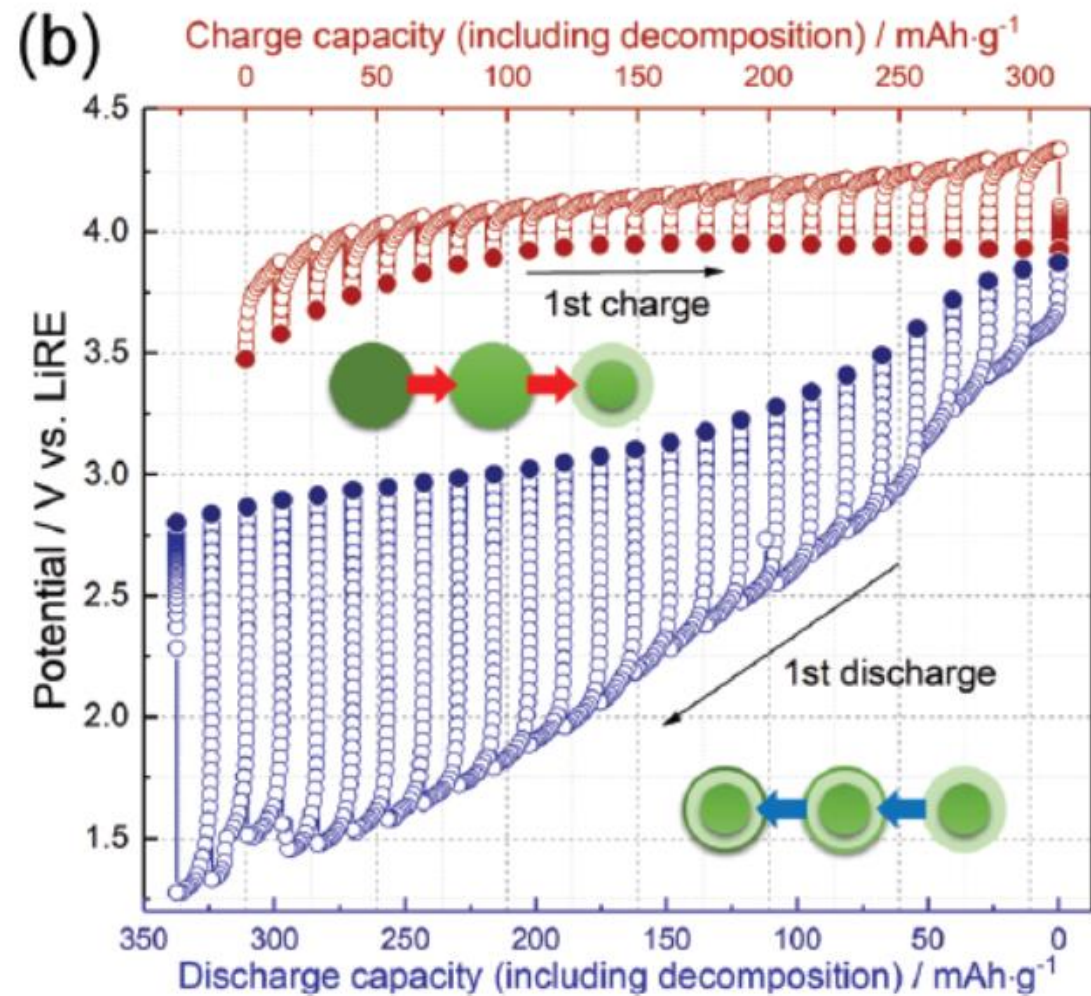
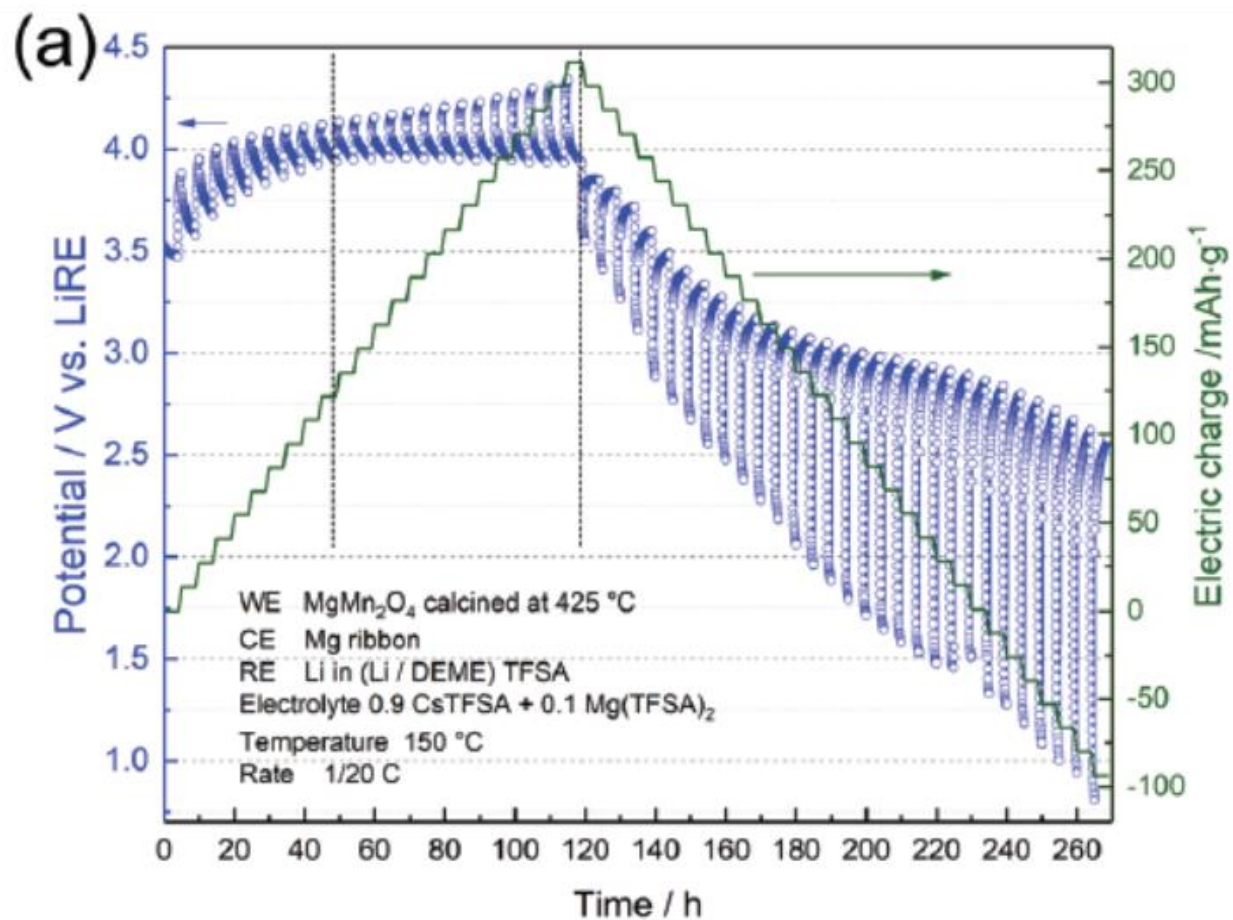
(c)



MgMn₂O₄からのMgの脱離挙動について



MgMn₂O₄からのMgの脱離挙動について





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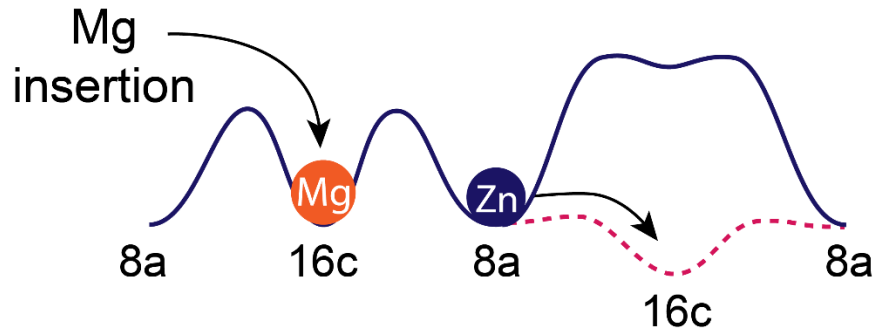
Cite this: *Phys. Chem. Chem. Phys.*,
2019, 21, 23749

Electrochemical phase transformation accompanied with Mg extraction and insertion in a spinel MgMn_2O_4 cathode material†

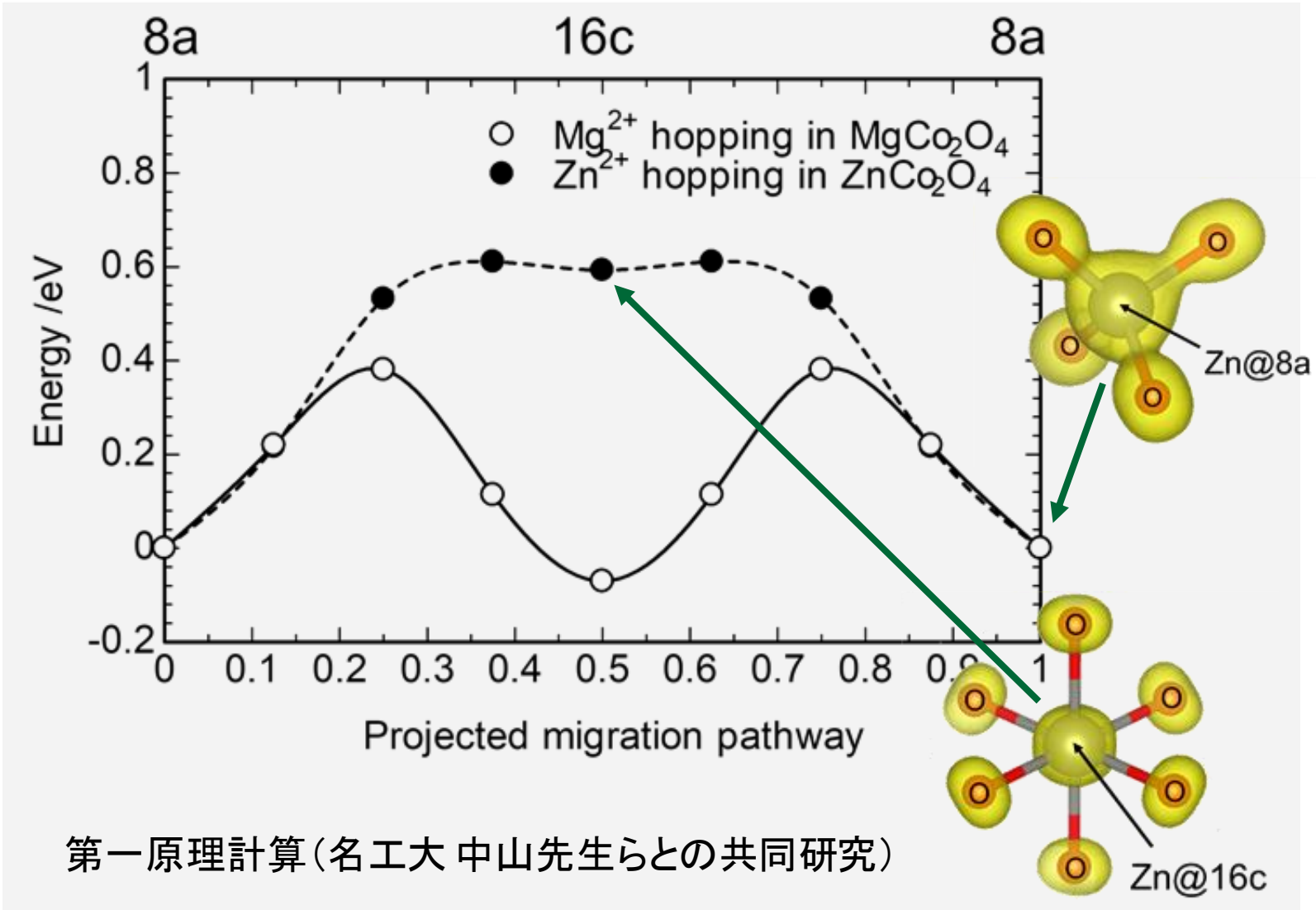
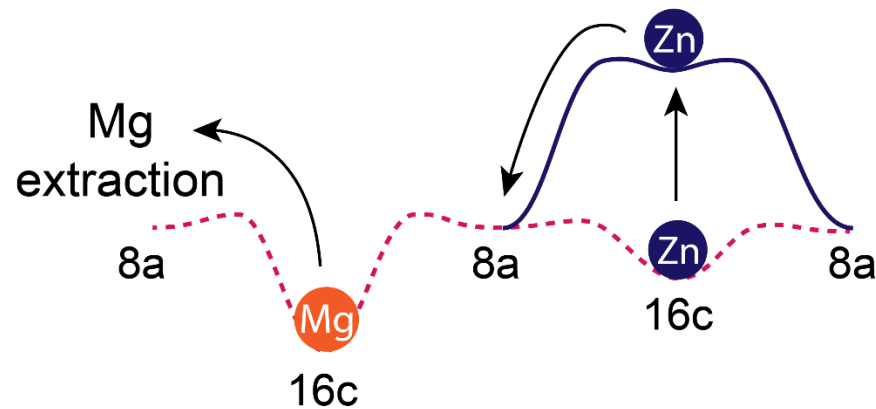
Takuya Hatakeyama, *^a Norihiko L. Okamoto, ^a Kohei Shimokawa, ^a
Hongyi Li, ^a Aiko Nakao, ^b Yoshiharu Uchimoto, ^c Hiroshi Tanimura,^a
Tomoya Kawaguchi ^a and Tetsu Ichitsubo *^a

四面体配位を好むZnの利用による「岩塩→スピネル」逆変態の促進

Discharge: spinel → rocksalt



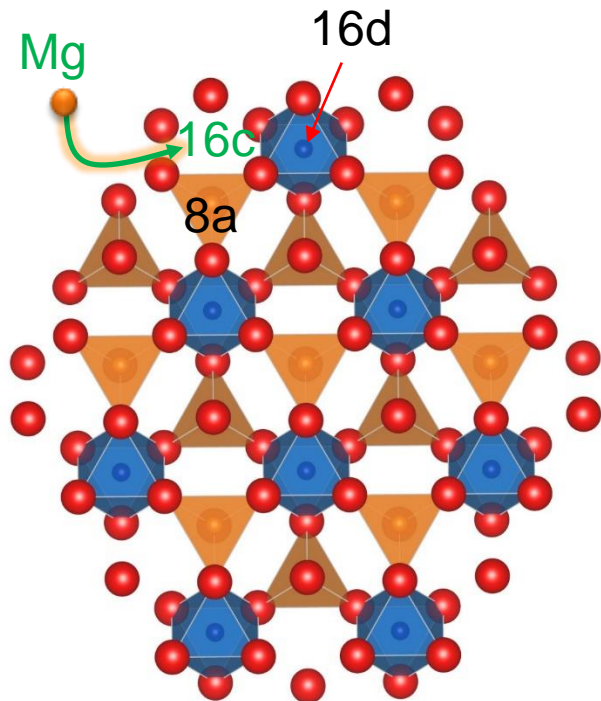
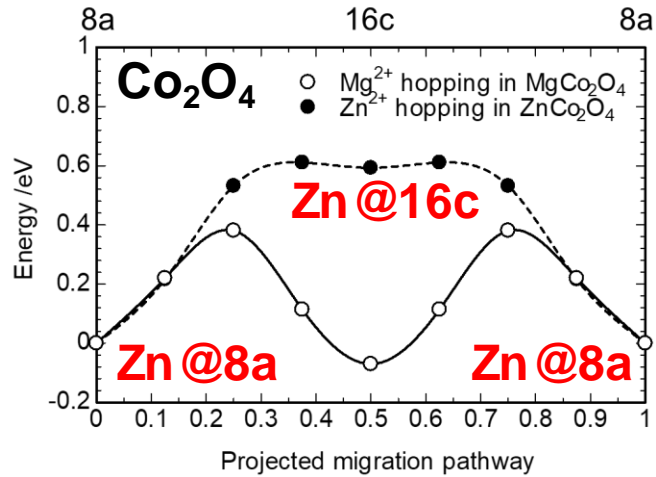
Charge: rocksalt → spinel



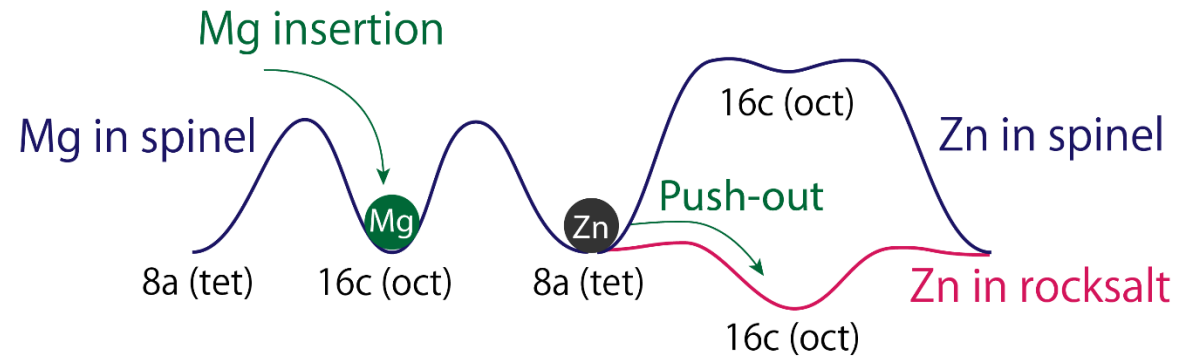
第一原理計算(名工大 中山先生らとの共同研究)

Znは充電時に「16c→8a」の移動(スピネル型構造への復帰)が速やかに生じると期待される。

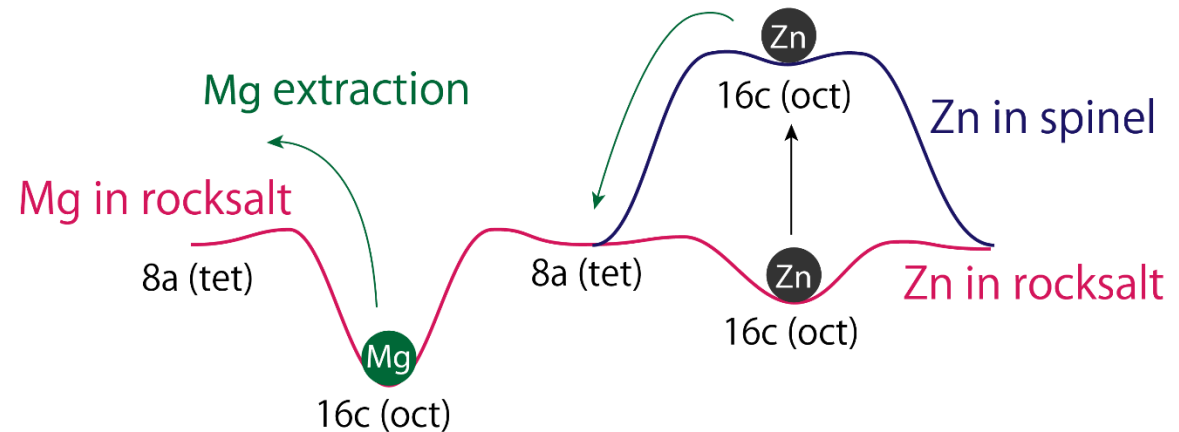
四面体配位を好むZnの利用



Discharge: spinel → rocksalt

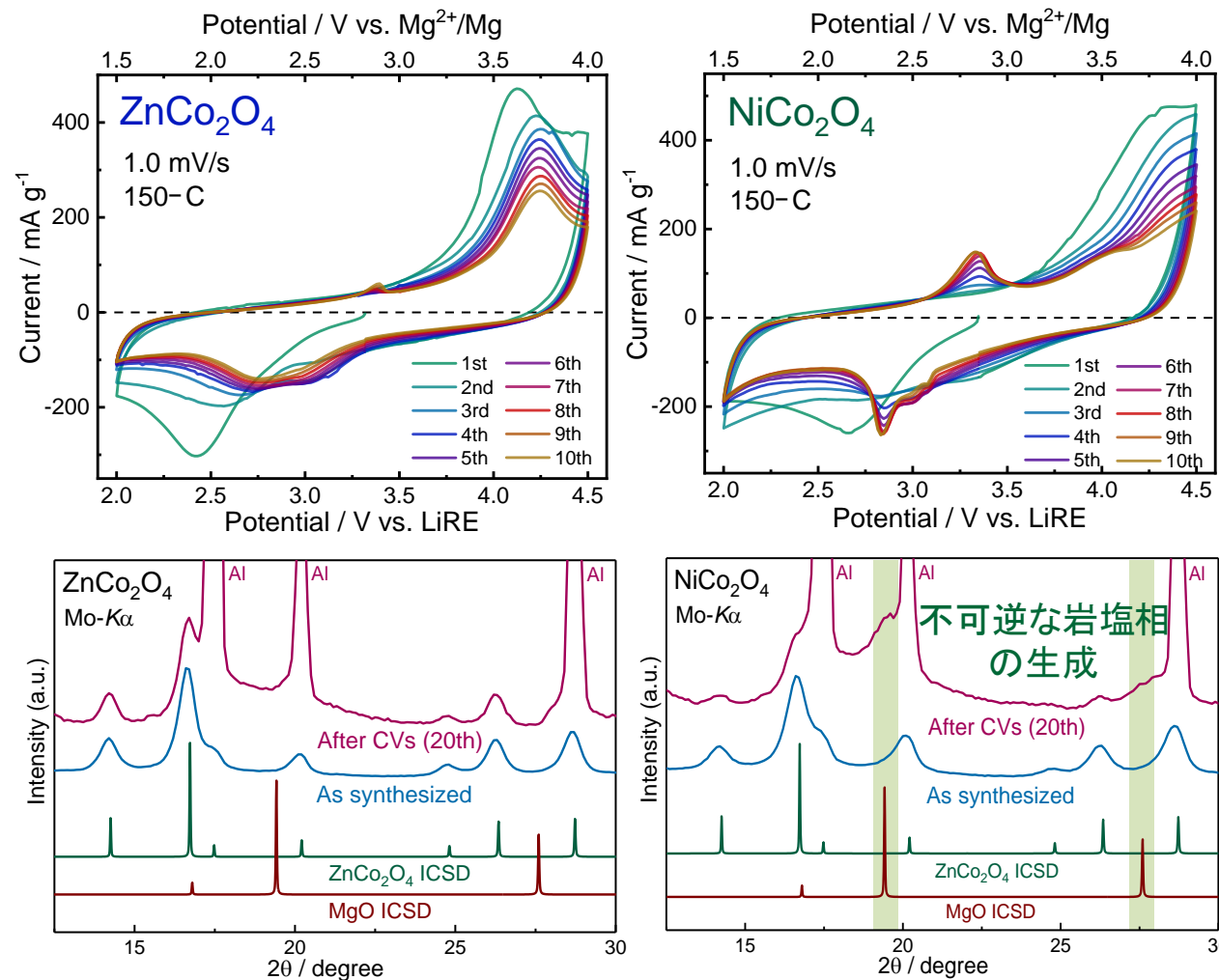


Charge: rocksalt → spinel

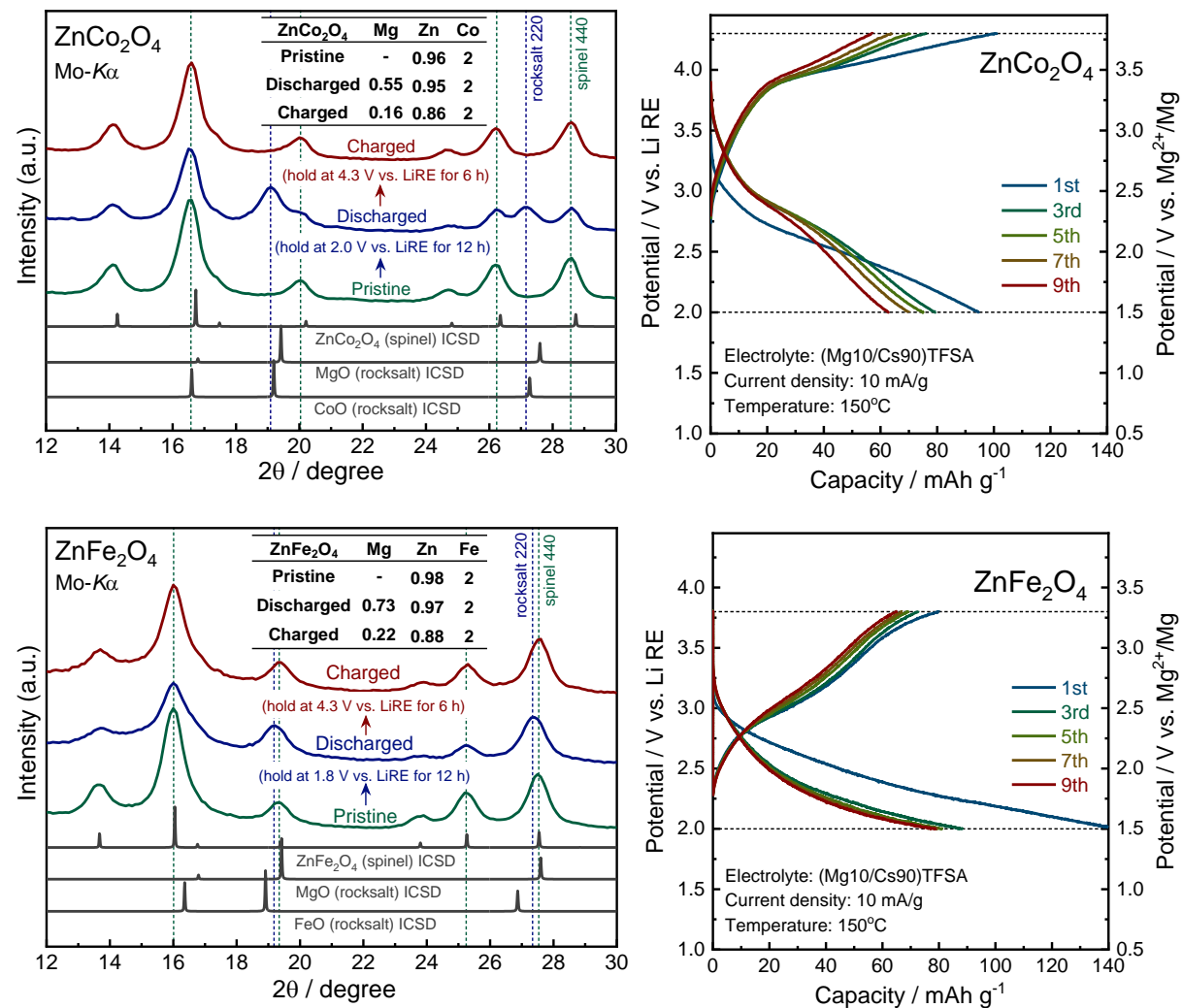


A(II)B(III)₂O₄ (A: Zn, Ni; B: Co, Fe) の比較実験結果

Znの利用による「岩塩→スピネル」逆変態の促進



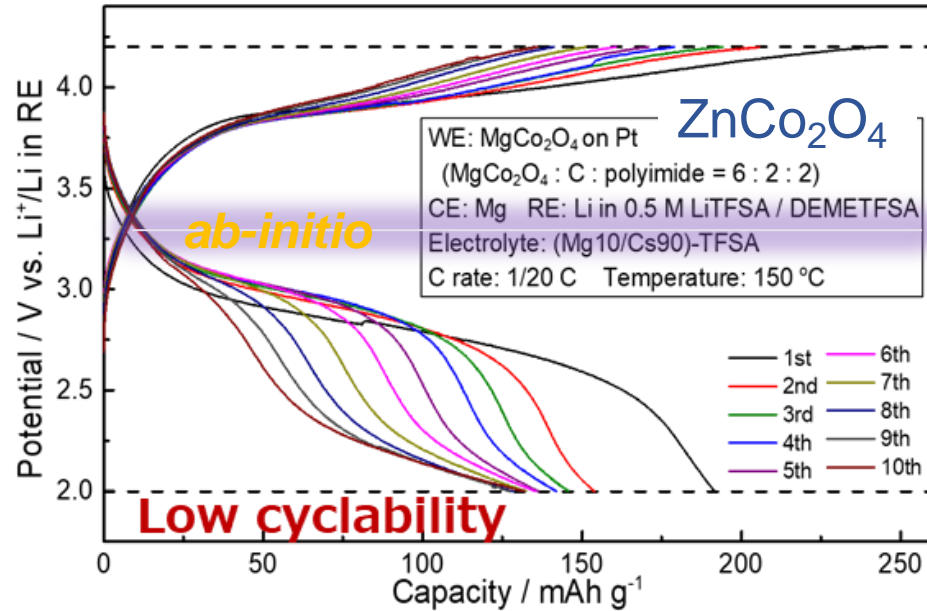
Feの利用による充放電時の体積変化率の低減



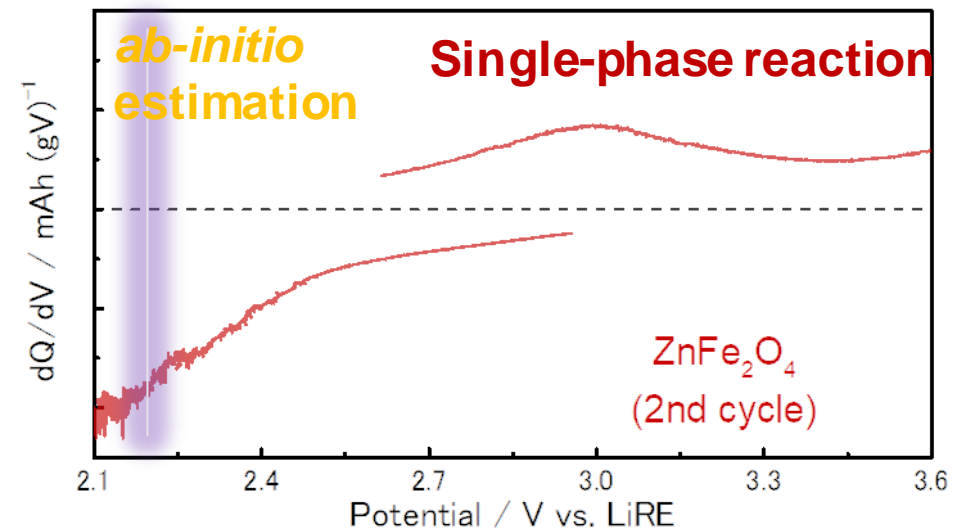
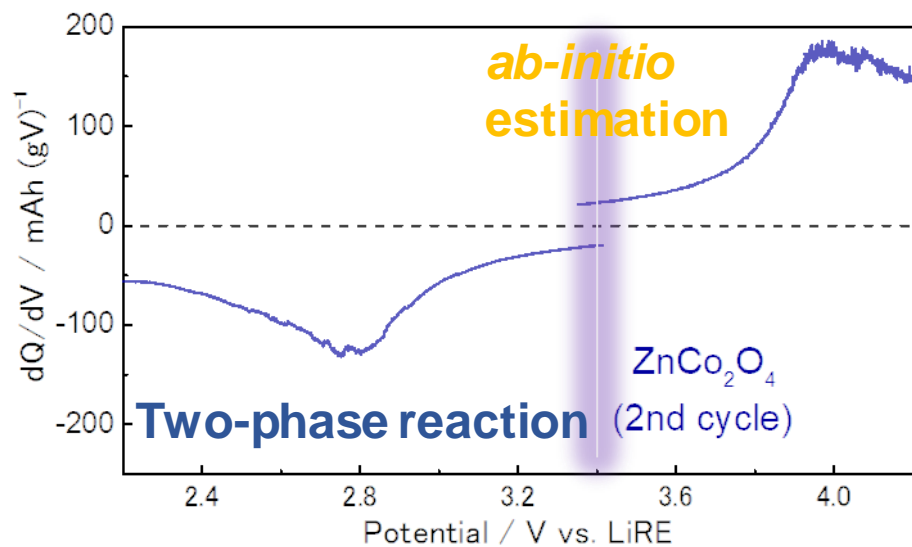
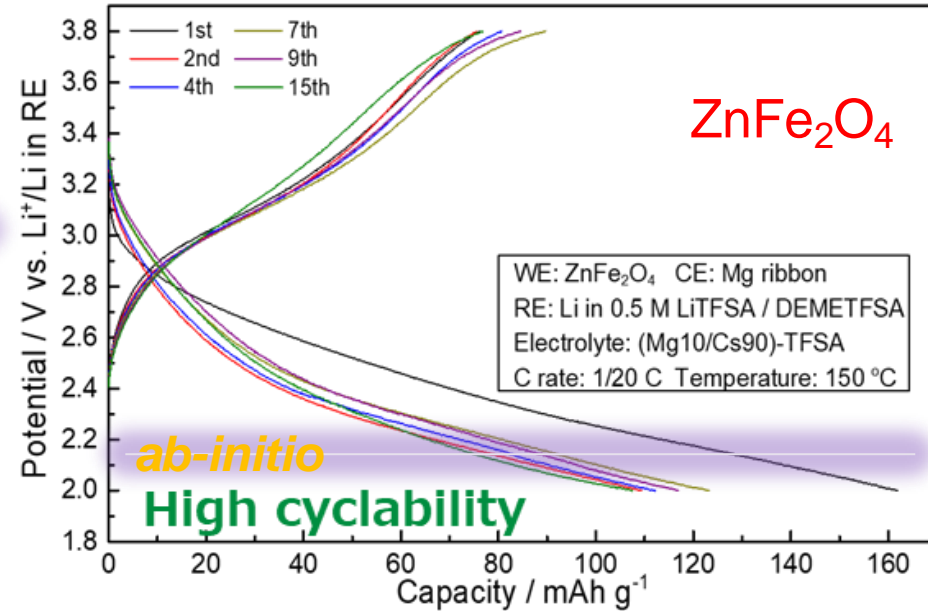
高サイクル特性の実現に向けたスピネル型酸化物の構造設計(カチオン選択)指針を構築

ZnCo₂O₄ vs ZnFe₂O₄ in cyclability

Large volume expansion and contraction



Almost zero strain



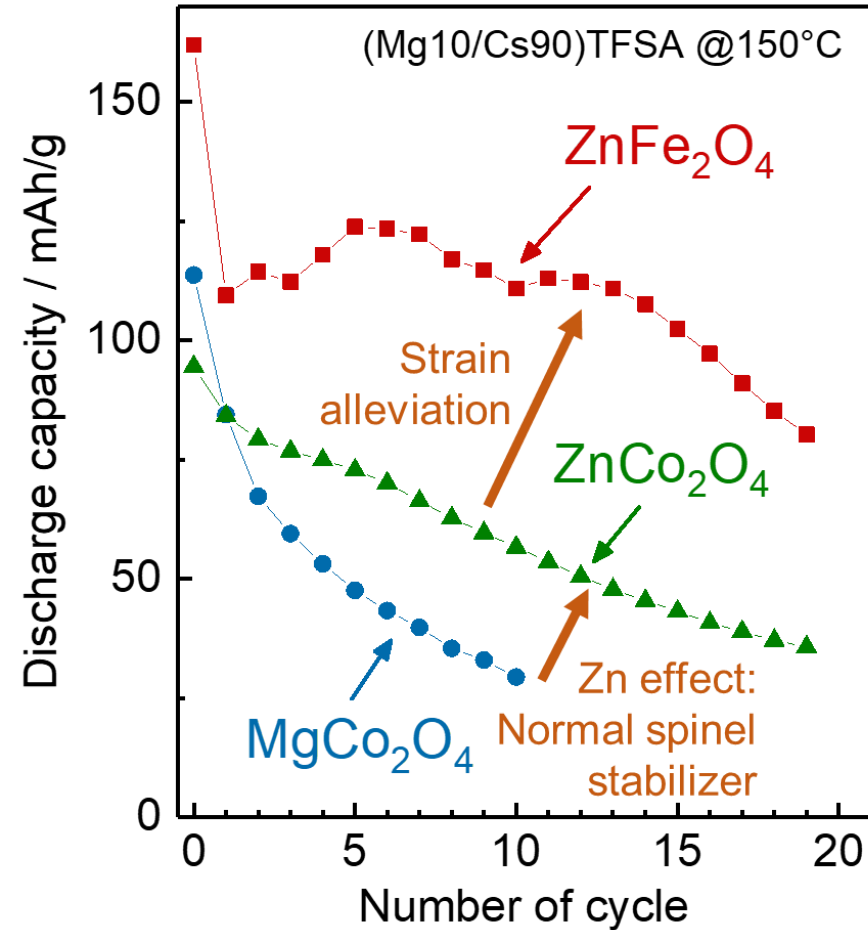
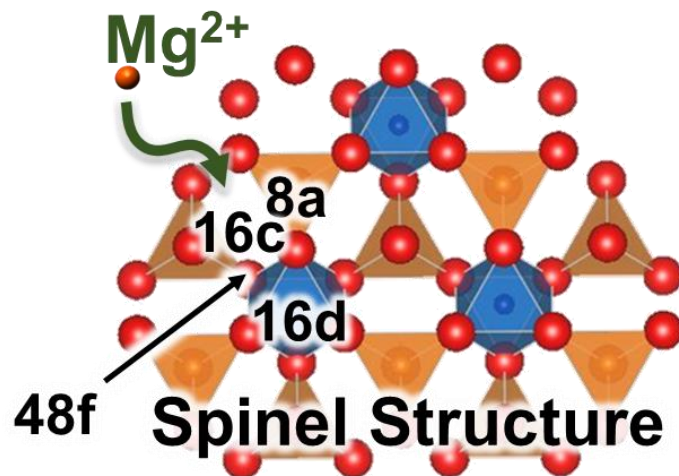
Materials design for MRB cathode

Spinel \leftrightarrow Rocksalt

Evidence of Mg insertion

1. Two-phase reaction
2. In spinel, 8a instead of 16c
In rocksalt, 16c instead of 8a
3. Coherent strain is inevitably caused

Degrade of cyclability



Although cyclability is markedly improved, the degrade of cyclability is still observed. How to circumvent this phenomenon?

K. Shimokawa, T. Atsumi, M. Harada, R. E. Ward, M. Nakayama, Y. Kumagai, F. Oba, N. L. Okamoto, K. Kanamura, T. Ichitsubo, [Journal of Materials Chemistry A 7, 12225-12235 \(2019\).](#)

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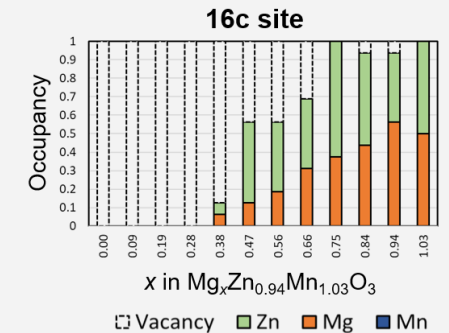
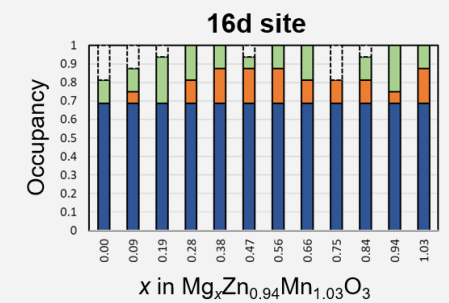
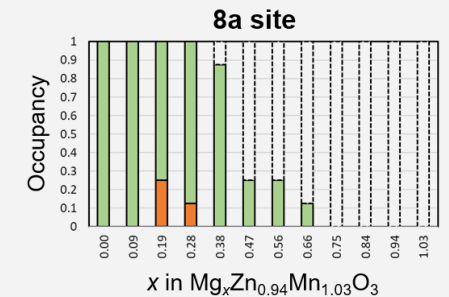
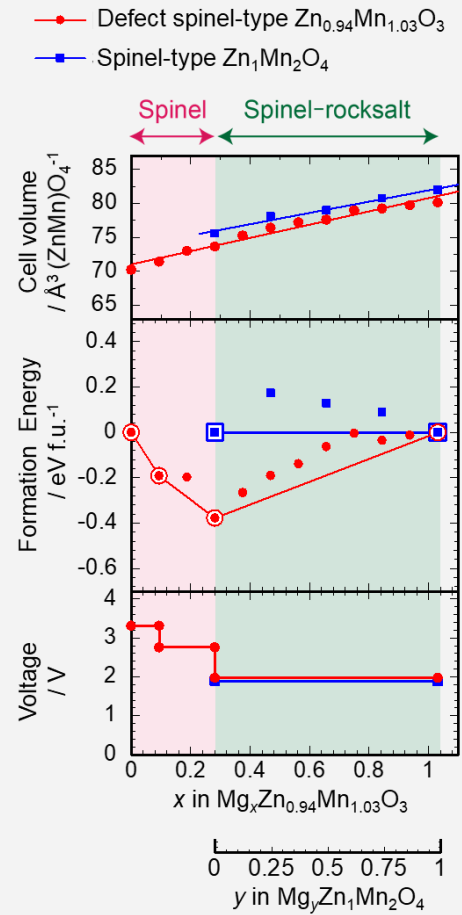
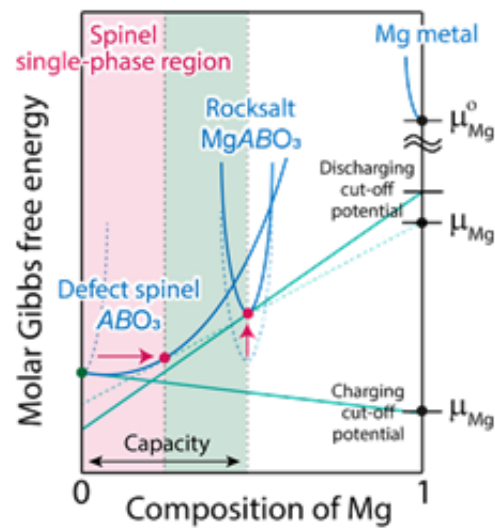
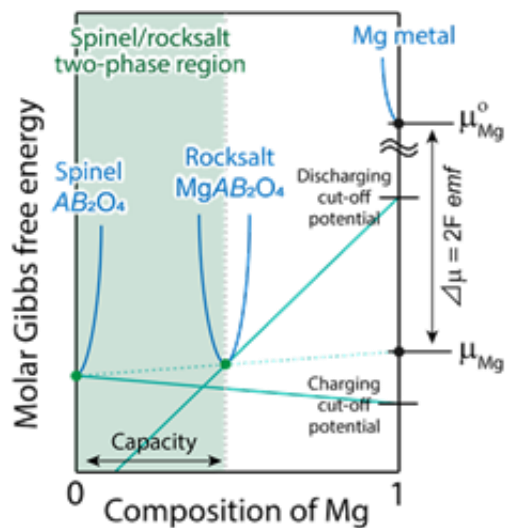
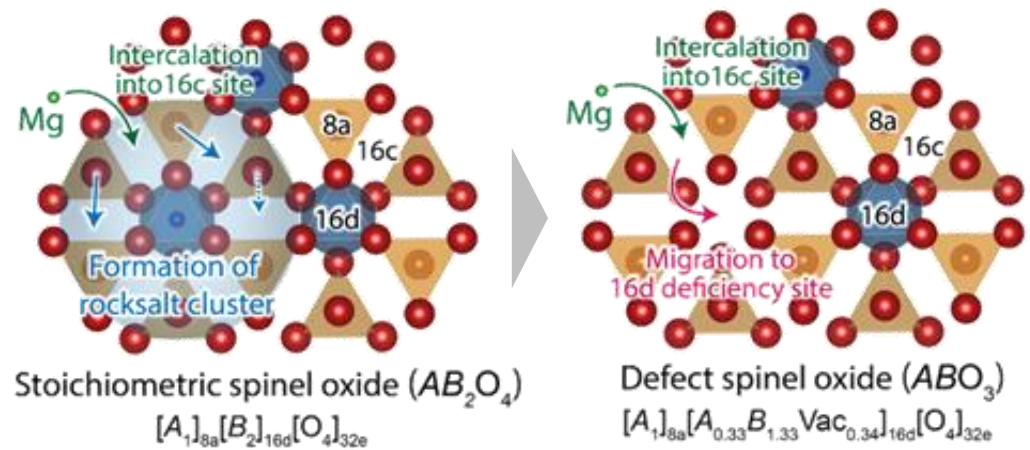
Check for updates

Cite this: *J. Mater. Chem. A*, 2019, 7, 12225

Zinc-based spinel cathode materials for magnesium rechargeable batteries: toward the reversible spinel–rocksalt transition†

Kohei Shimokawa, *^a Taruto Atsumi,^b Maho Harada,^b Robyn E. Ward,^c Masanobu Nakayama, ^{bc} Yu Kumagai,^d Fumiyasu Oba,^{ce} Norihiko L. Okamoto, ^a Kiyoshi Kanamura^f and Tetsu Ichitsubo *^a

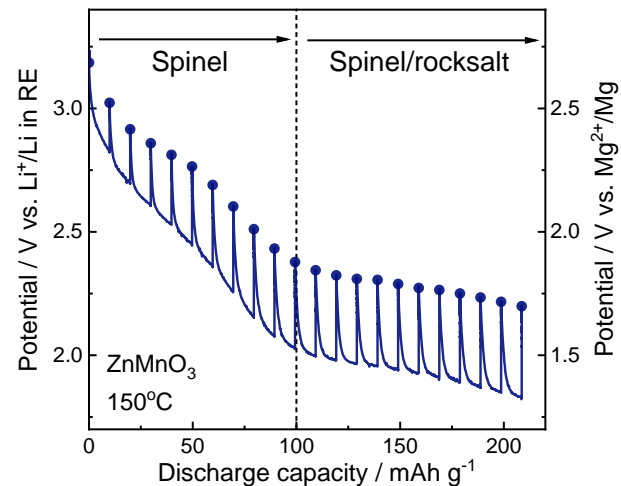
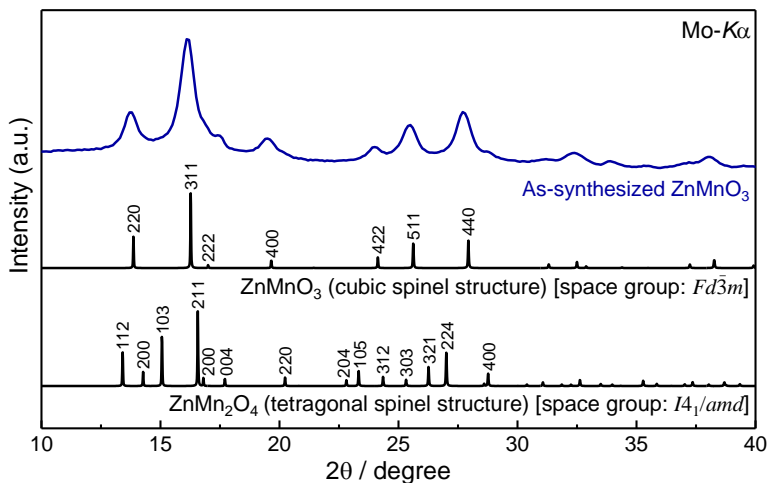
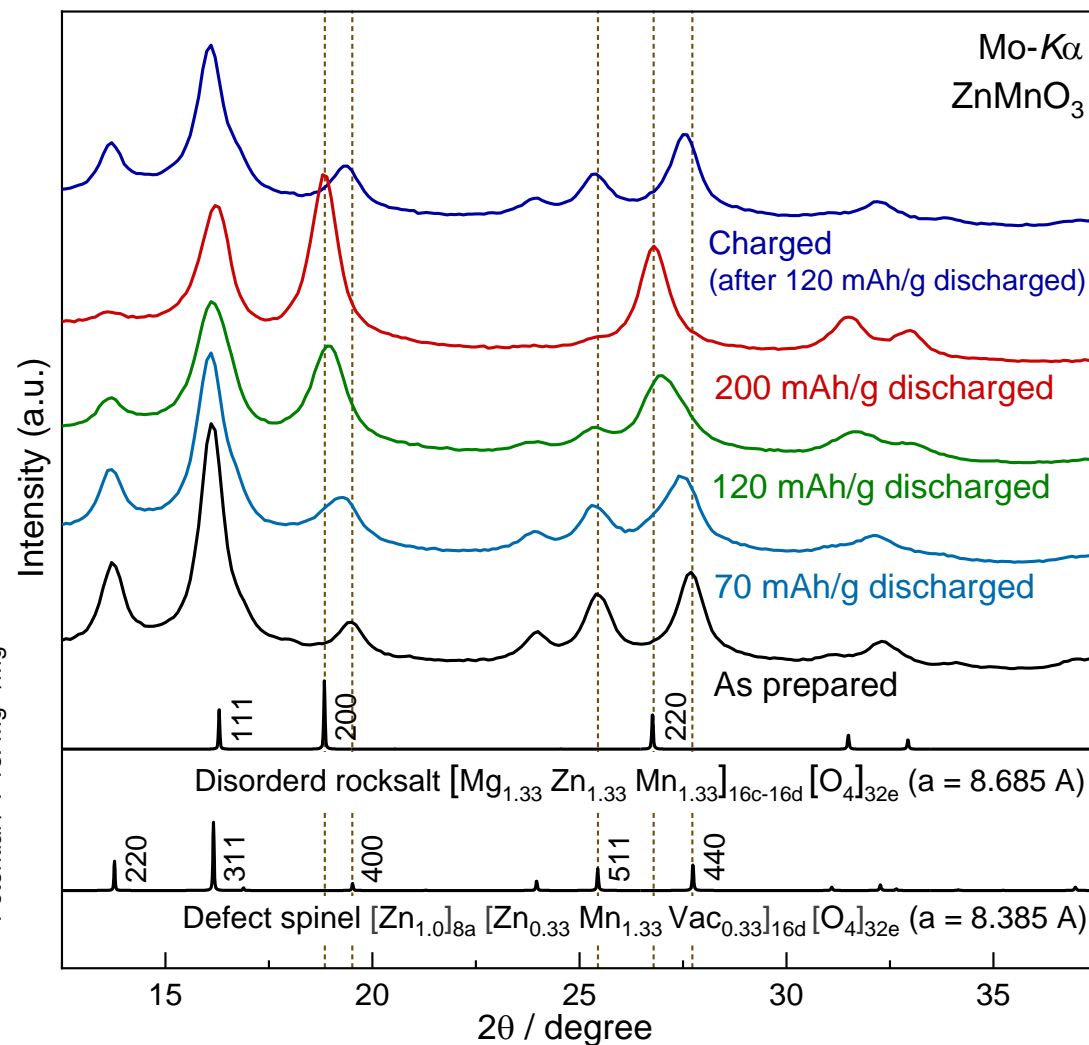
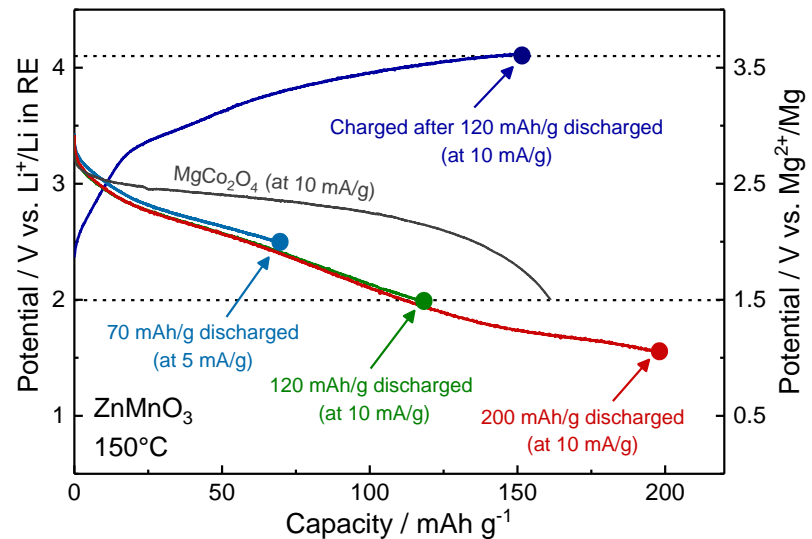
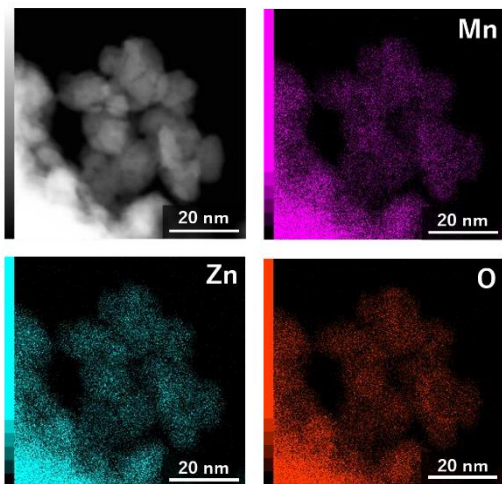
欠陥スピネル型酸化物 $ZnMnO_3$ の利用を提案



第一原理計算(名工大 中山先生らとの共同研究)

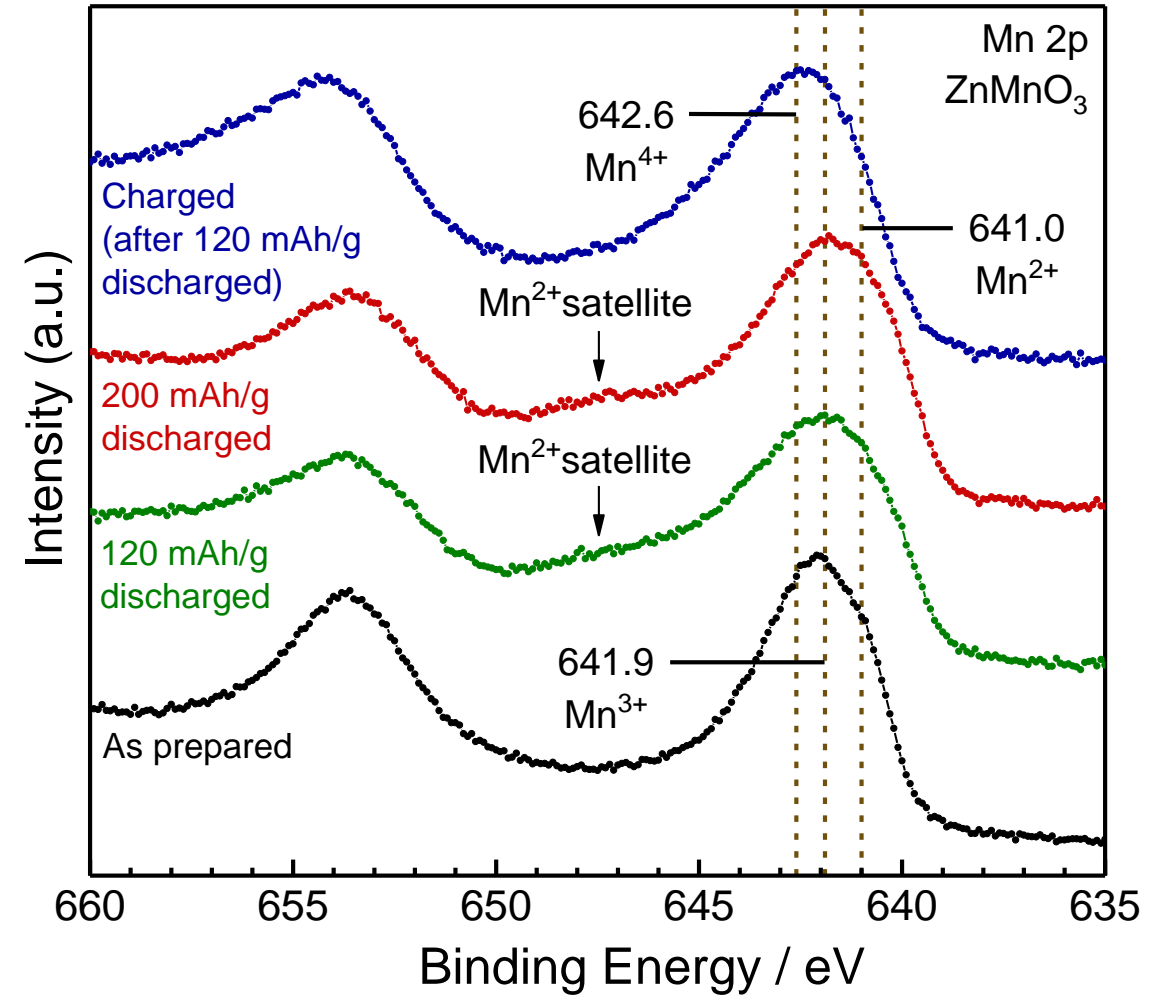
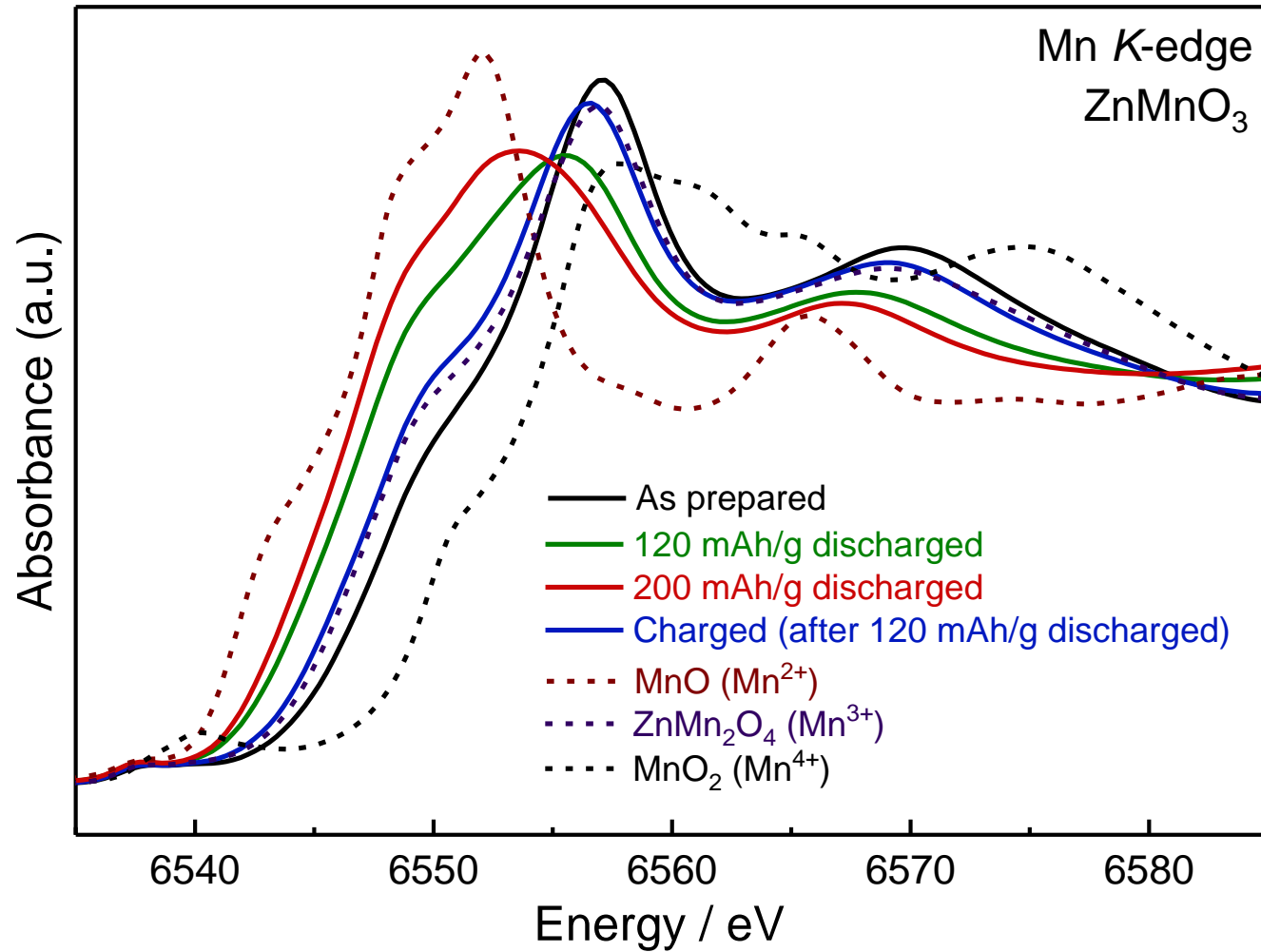
カチオン欠損サイトを利用することで岩塩相への二相共存反応による構造相転移を抑制し、サイクル特性の向上を狙う。

ZnMnO₃の合成と充放電挙動(初回サイクル)



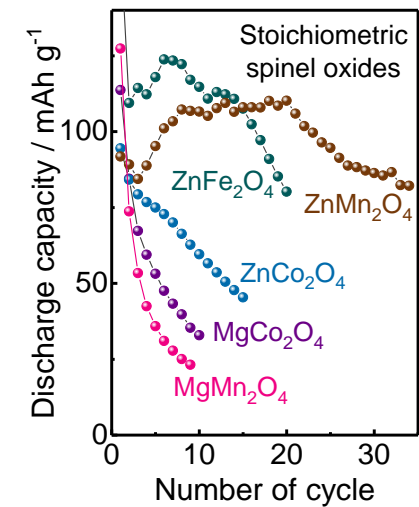
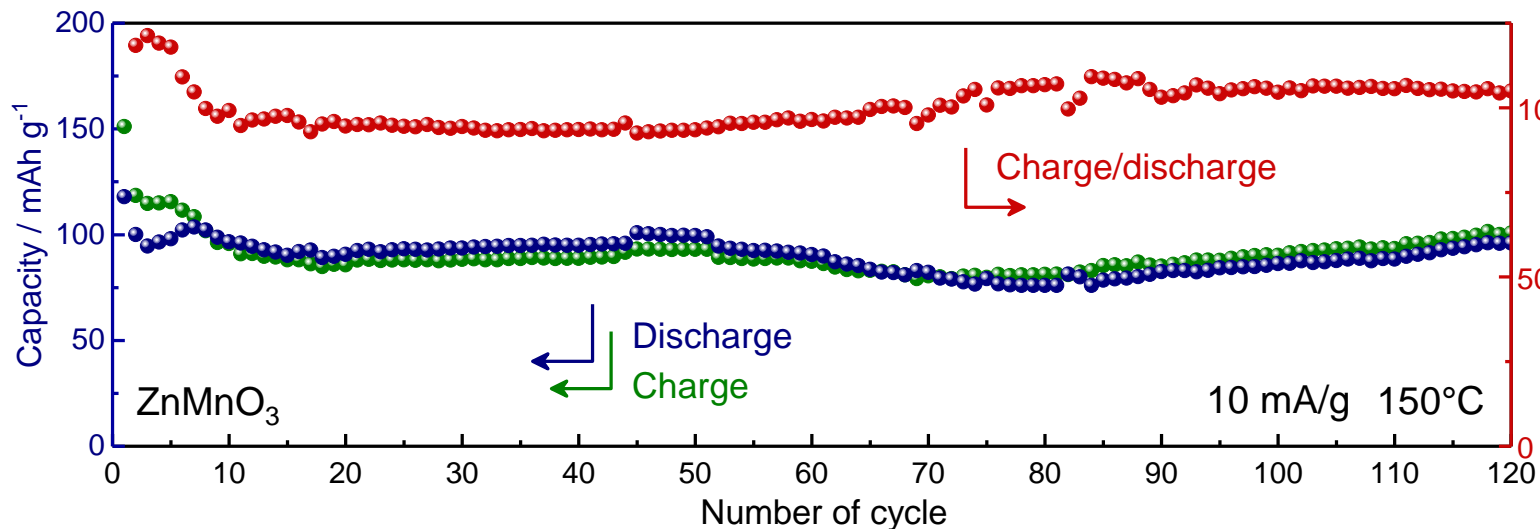
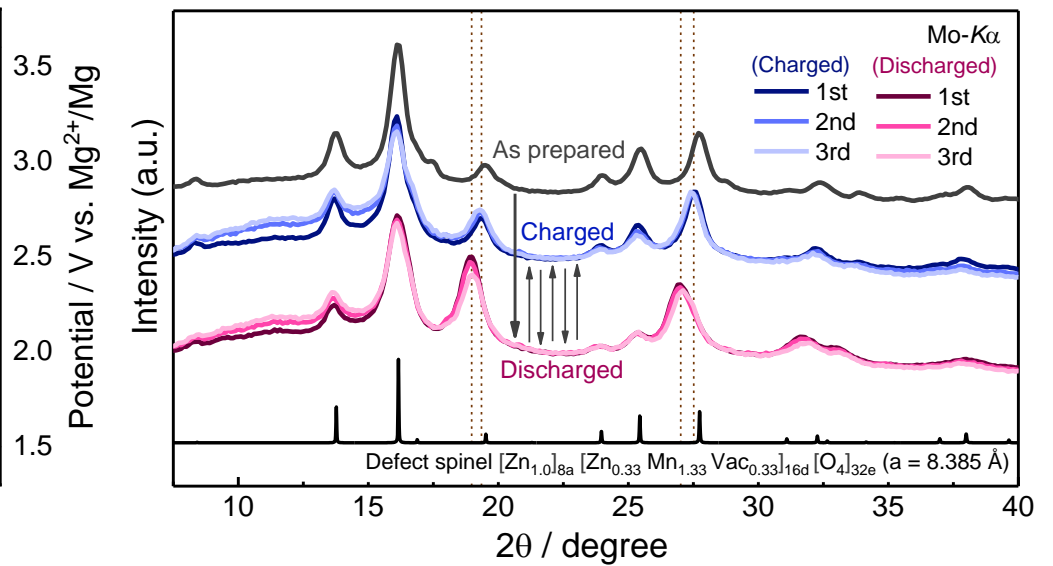
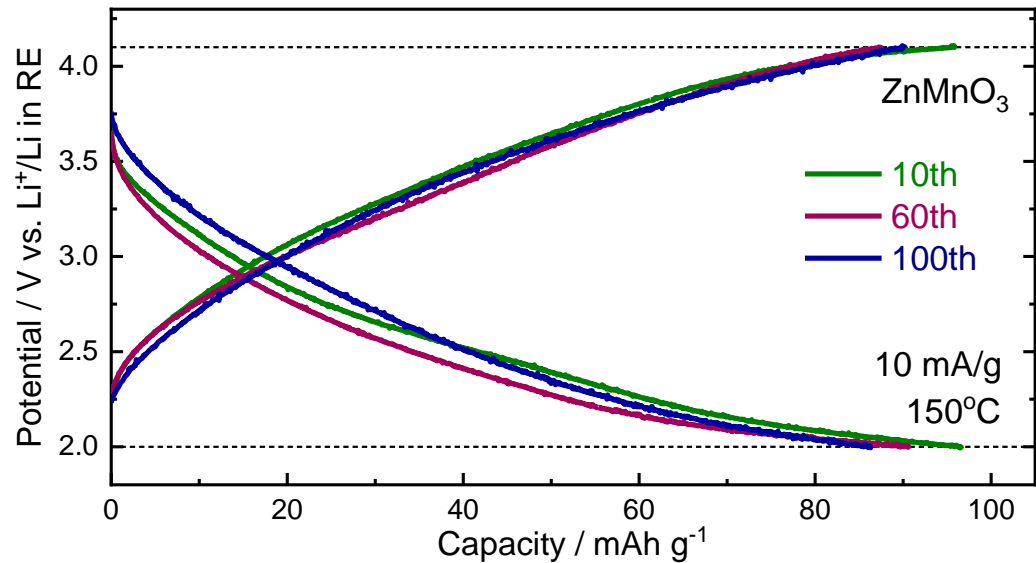
単相のZnMnO₃ナノ粒子の合成に成功. 放電初期の約100 mAh/gで単相反応を示唆.

充放電前後のMnの価数変化 (XANES, XPS)



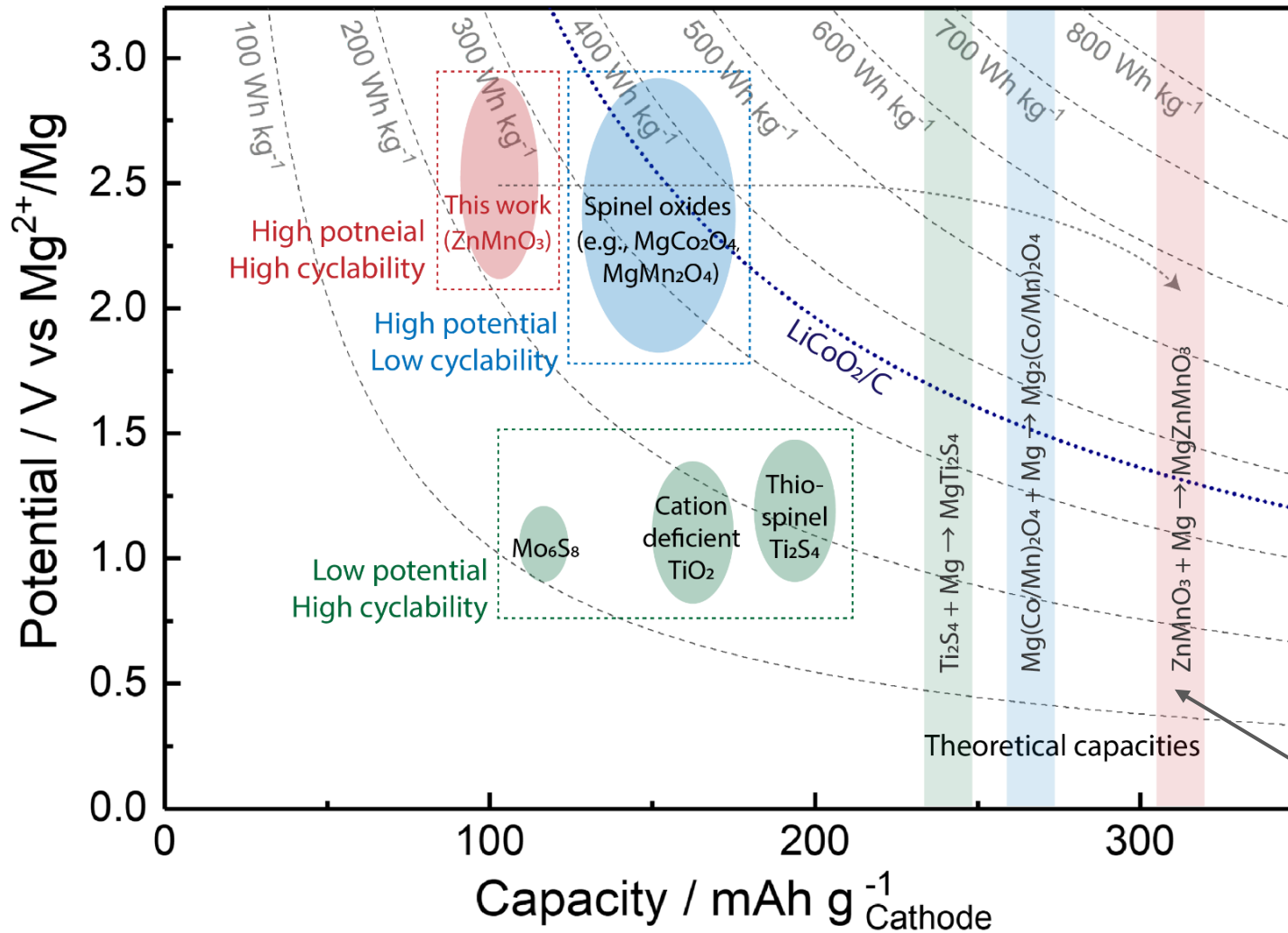
バルクと表面で多少の不均一性があるものの、可逆的なMnの価数変化を確認。

欠陥スピネルZnMnO₃の充放電試験結果(150°C)

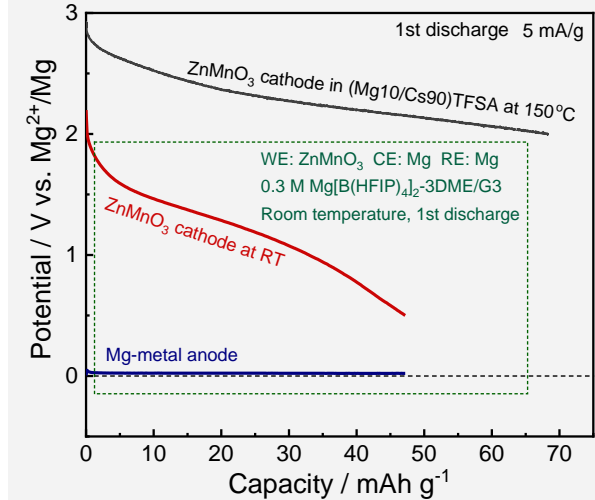


高電位・高容量・高サイクル特性を兼ね備えた正極特性を実現

Liイオン電池に迫る理論エネルギー密度



フルセル(室温)構築の試み



小林先生・万代先生よりご提供
いただいた $Mg[B(HFIP)_4]_2$ -
3DME/G3電解液を使用

※ スピネル/岩塩の二相共存
領域も含めた理論容量

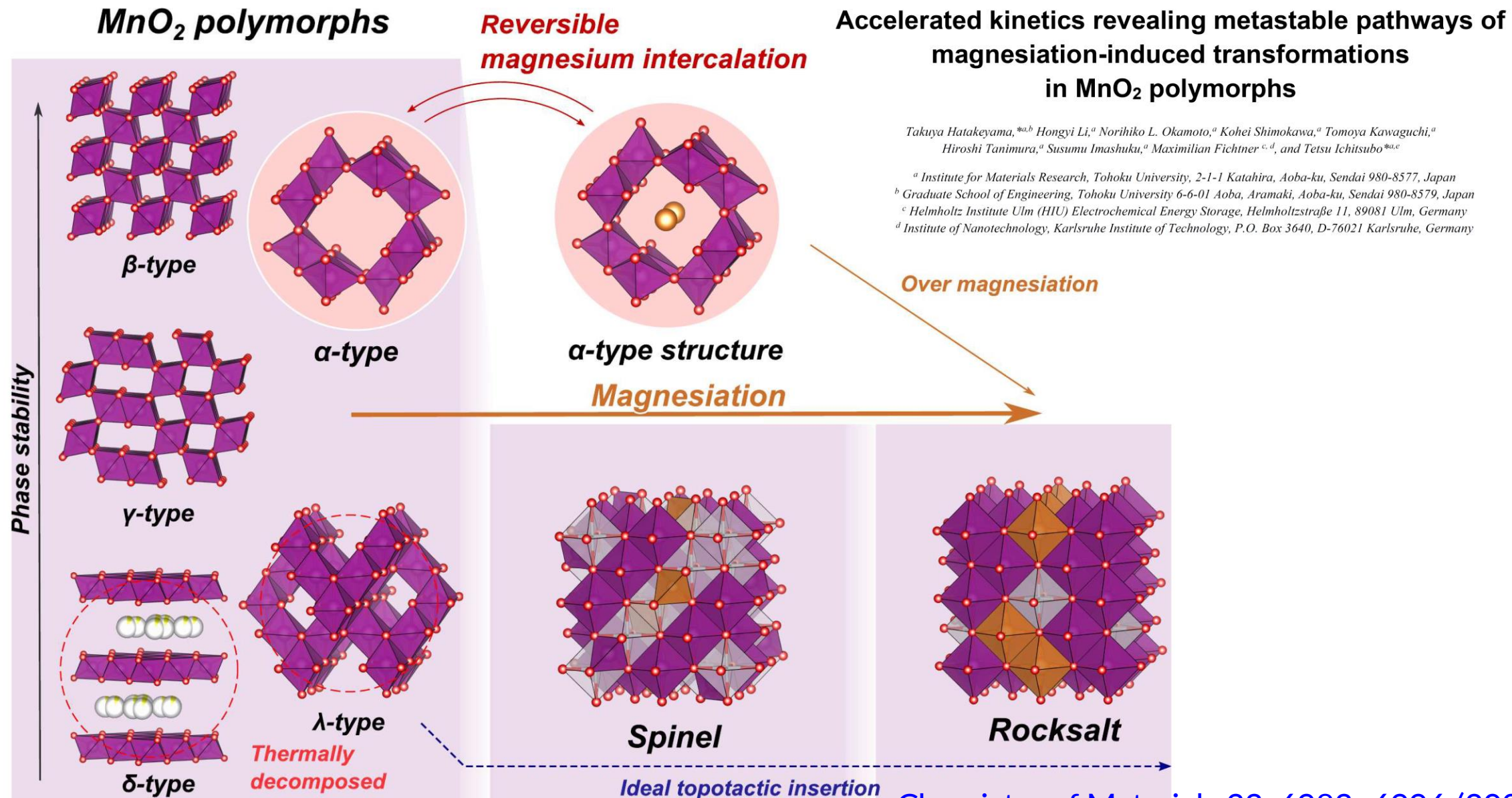
残された課題: ①作動温度の低減, ②正極/負極と適合性のある電解液の開発

Structure Design of Long-Life Spinel-Oxide Cathode Materials for Magnesium Rechargeable Batteries

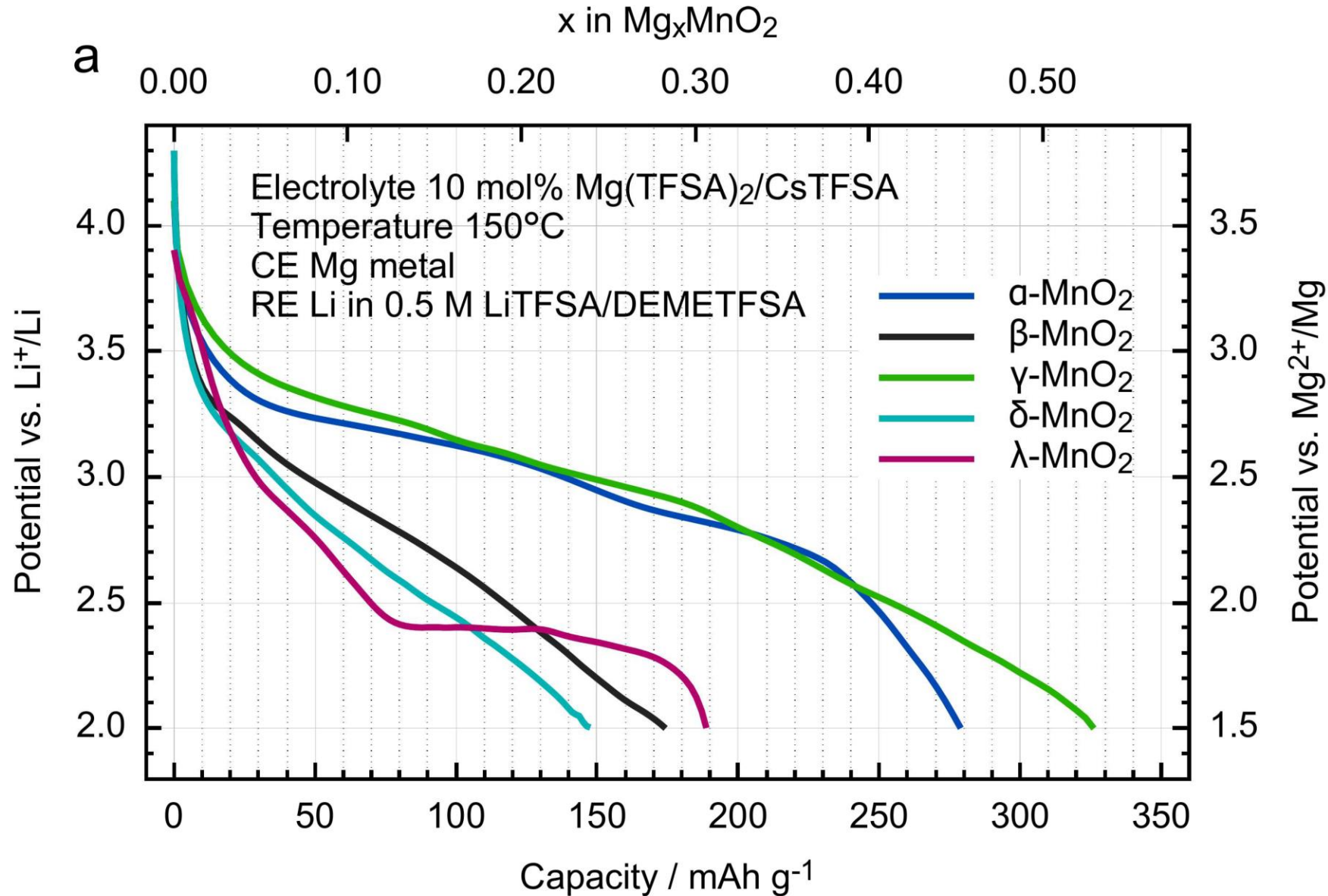
Kohei Shimokawa, Taruto Atsumi, Norihiko L. Okamoto, Tomoya Kawaguchi, Susumu Imashuku, Kazuaki Wagatsuma, Masanobu Nakayama, Kiyoshi Kanamura, and Tetsu Ichitsubo**

DOI: 10.1002/adma.202007539

MnO₂多形のMgイオン蓄電池正極に対する総合的研究

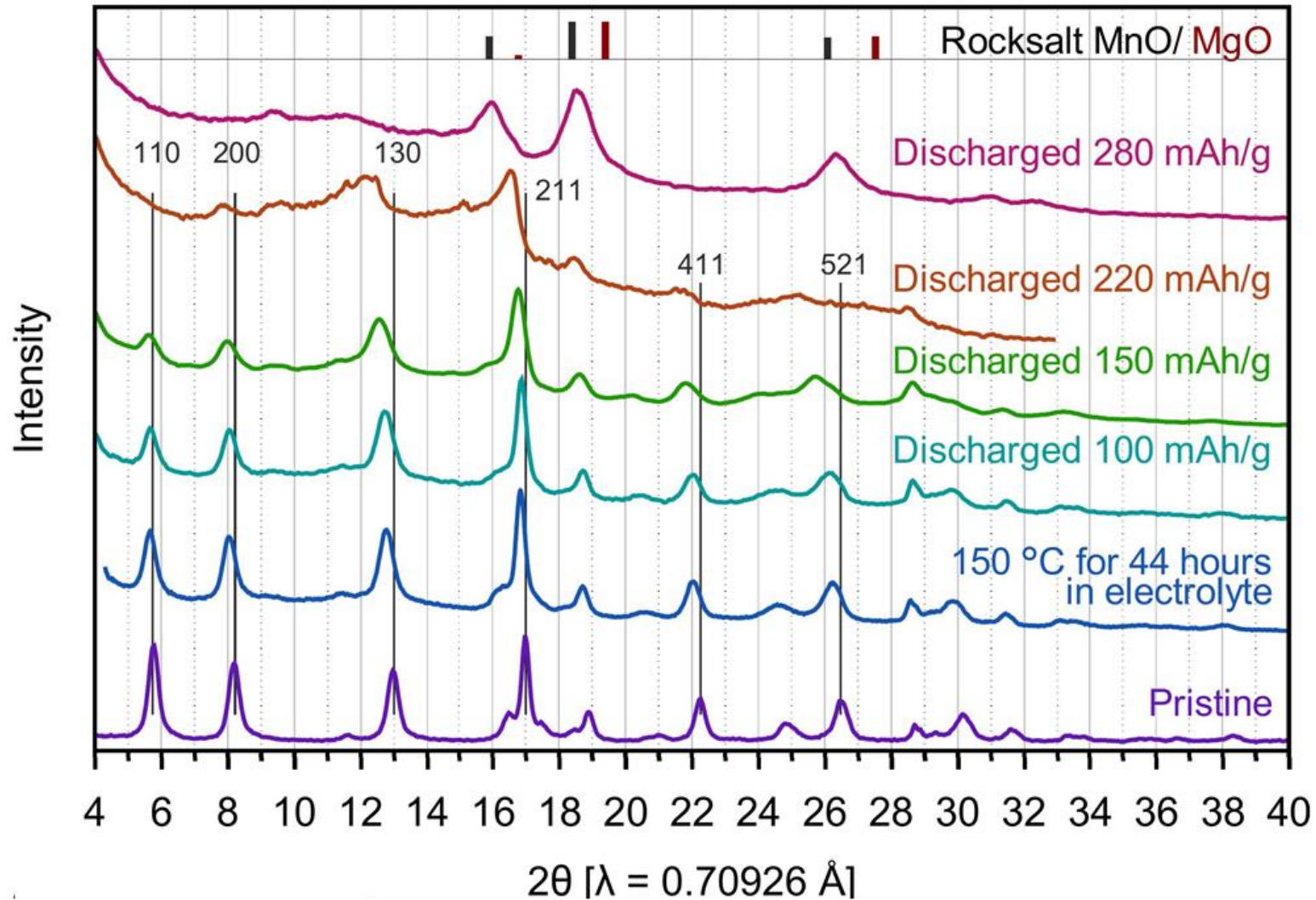


MnO₂多形のMagnesiation挙動

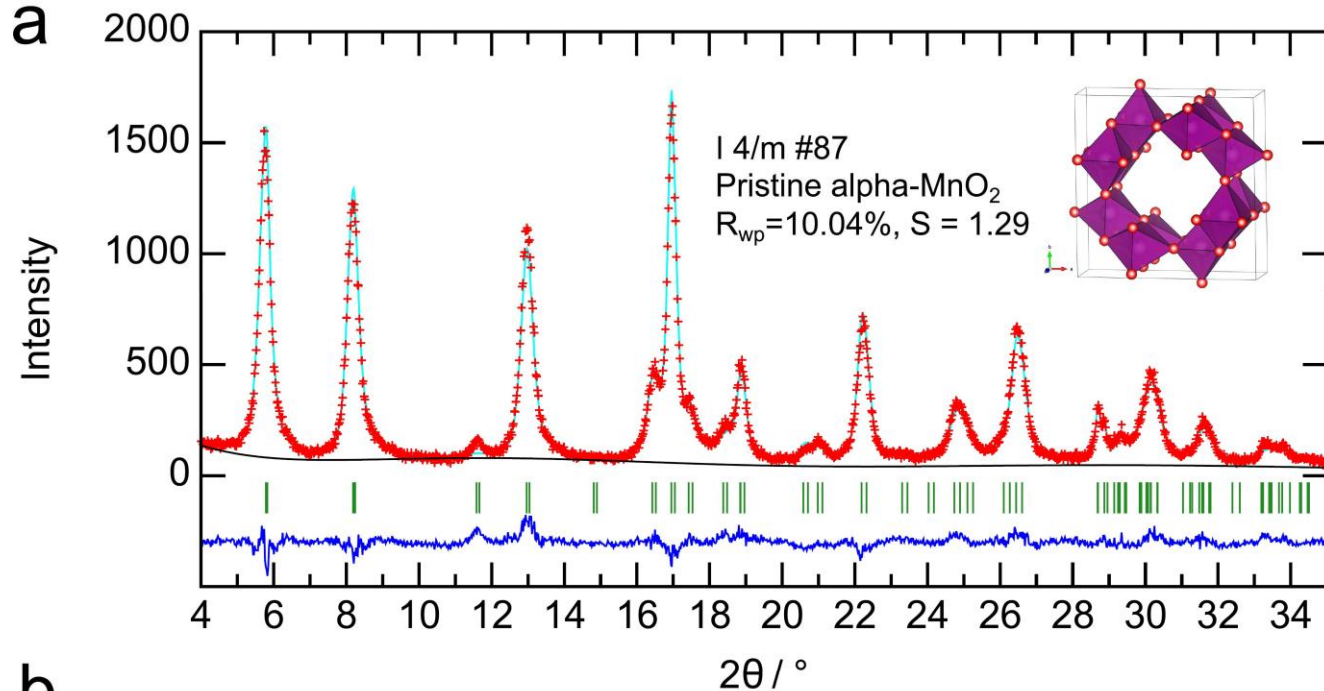


Magnesianation後の構造変化: α 型

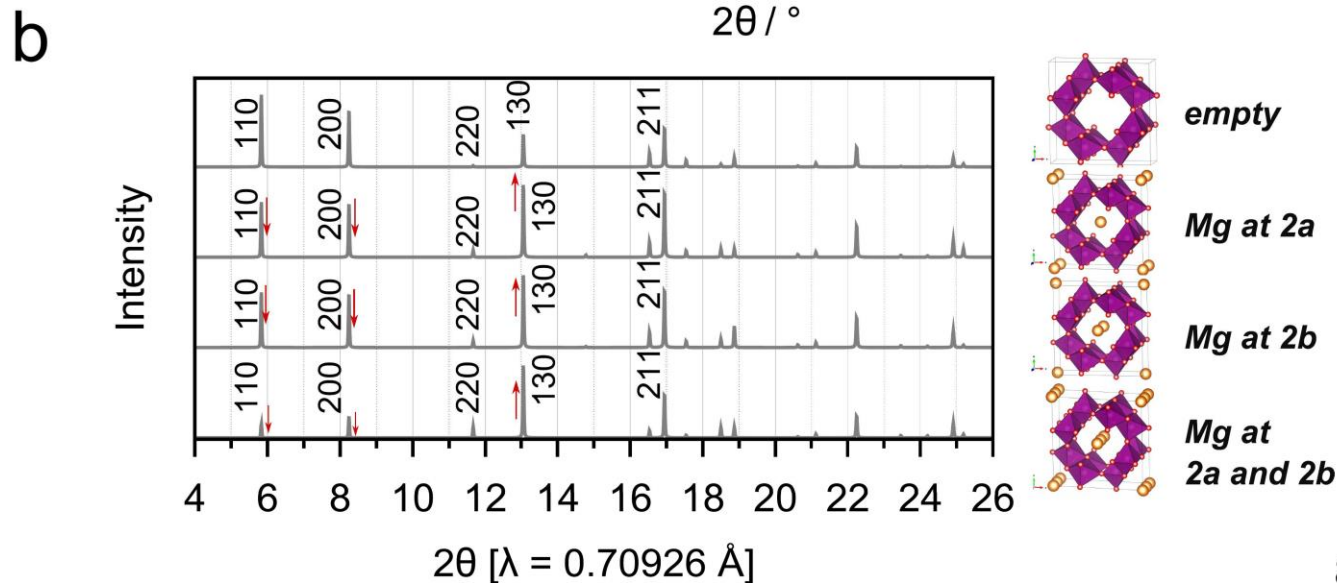
b α -MnO₂



Mgの挿入位置による構造変化: α 型

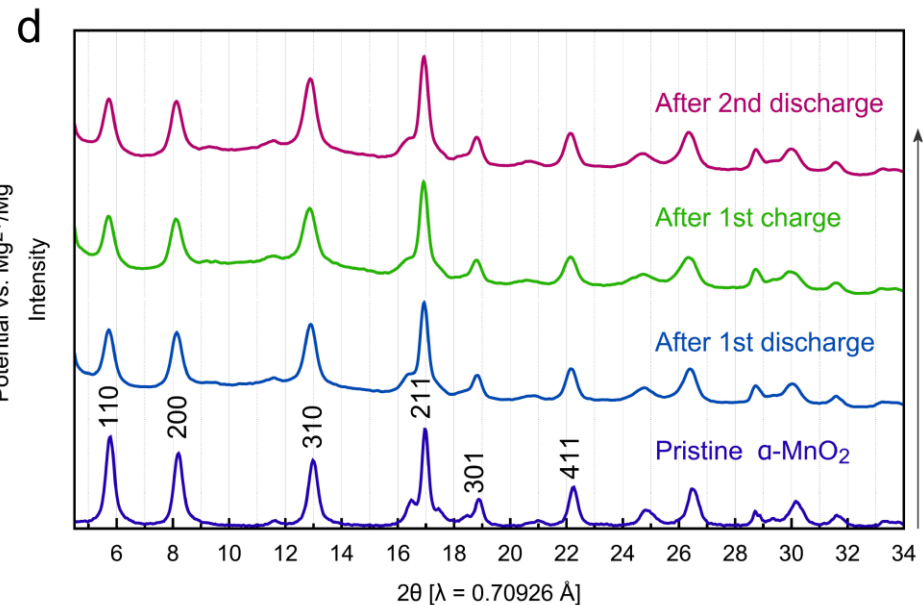
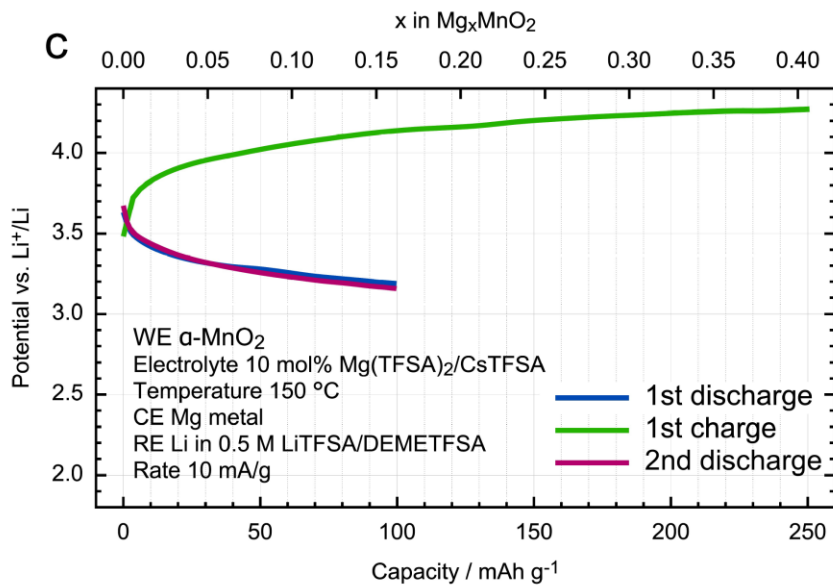
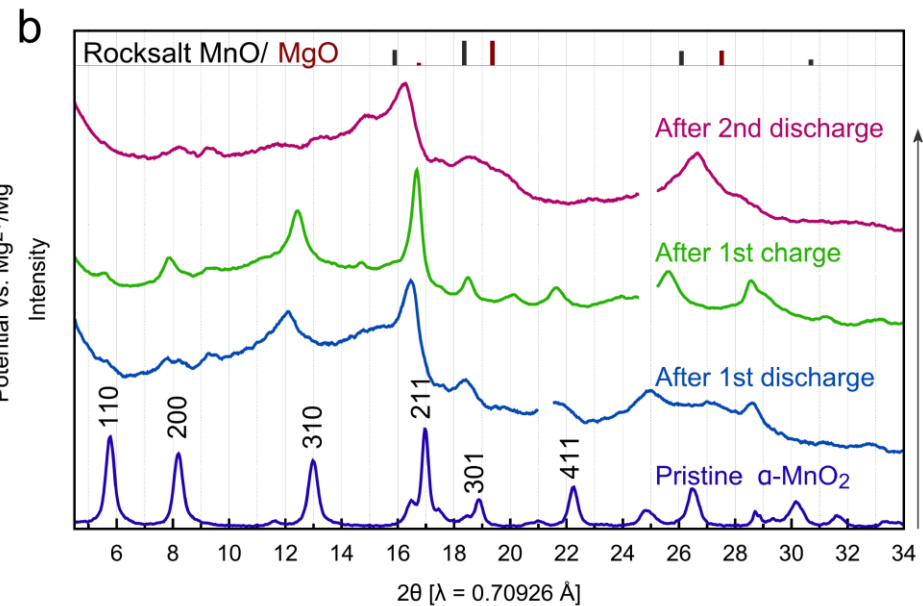
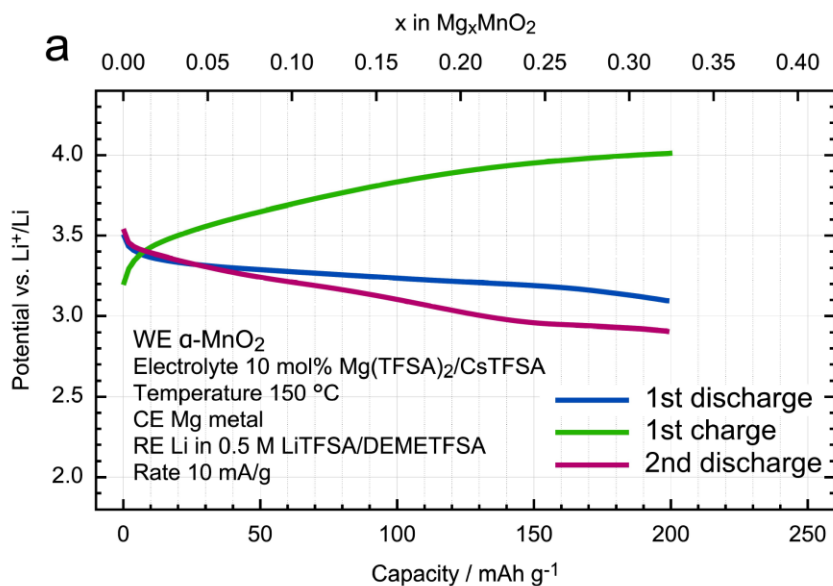


初期構造

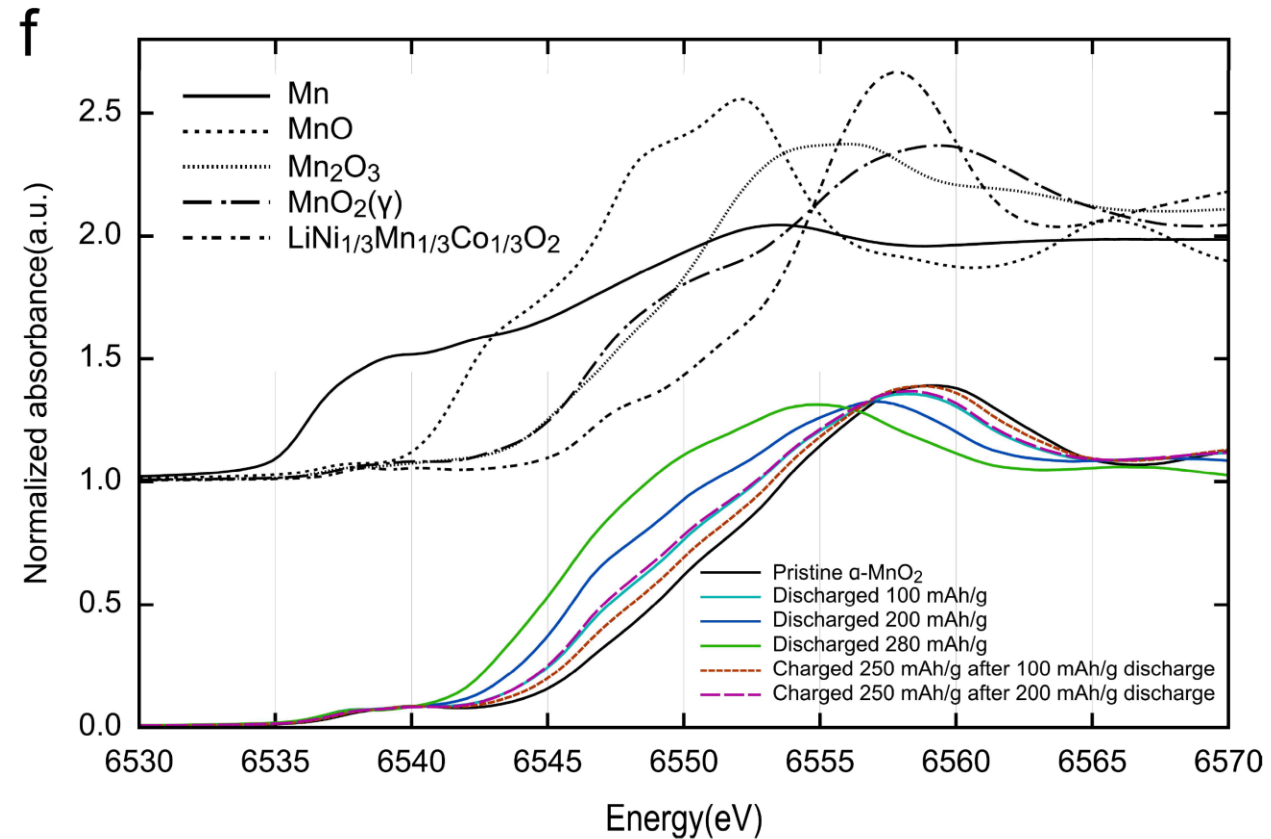
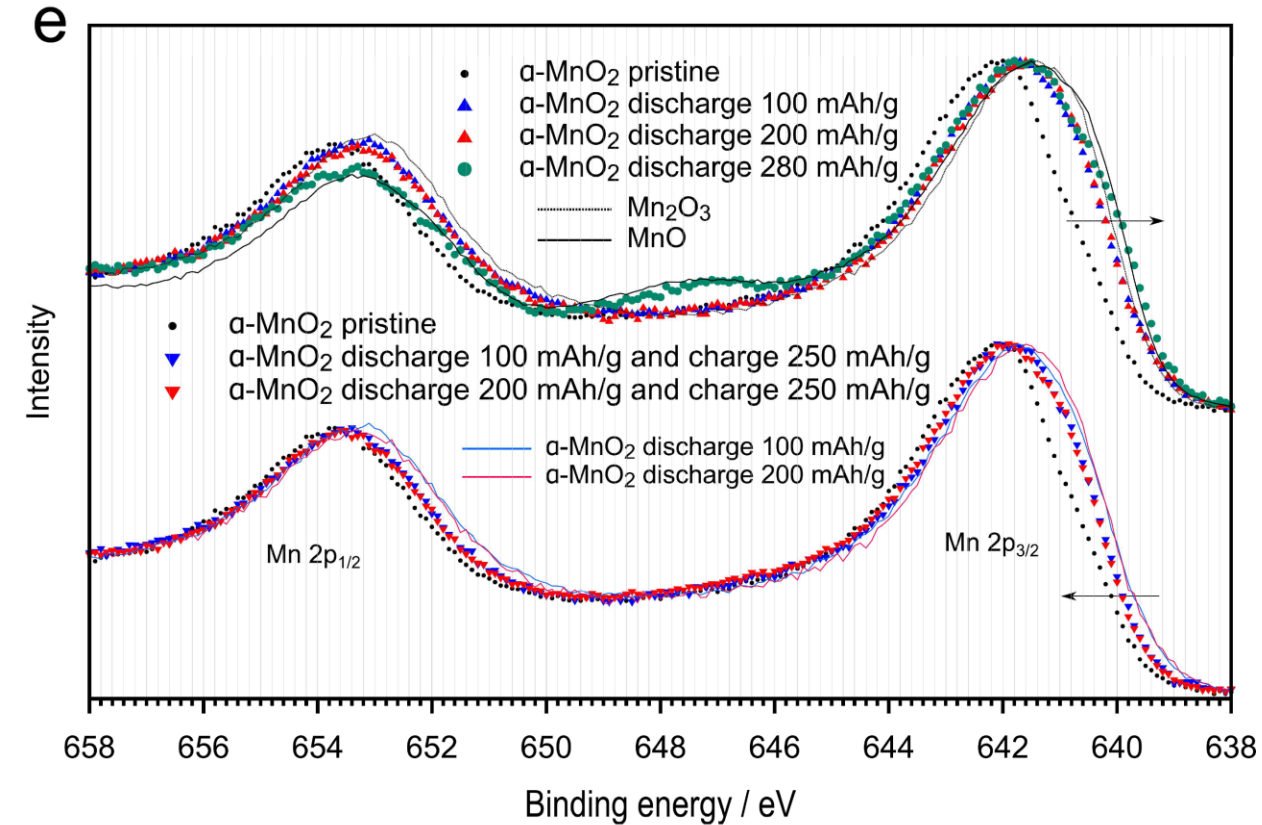


Mg挿入による
構造変化
シミュレーション

α 型のMg挿入・脱離実験と構造変化



α 型のMg挿入・脱離実験と価数変化 XPS & XANES



α 型のMg挿入脱離の相変態過程

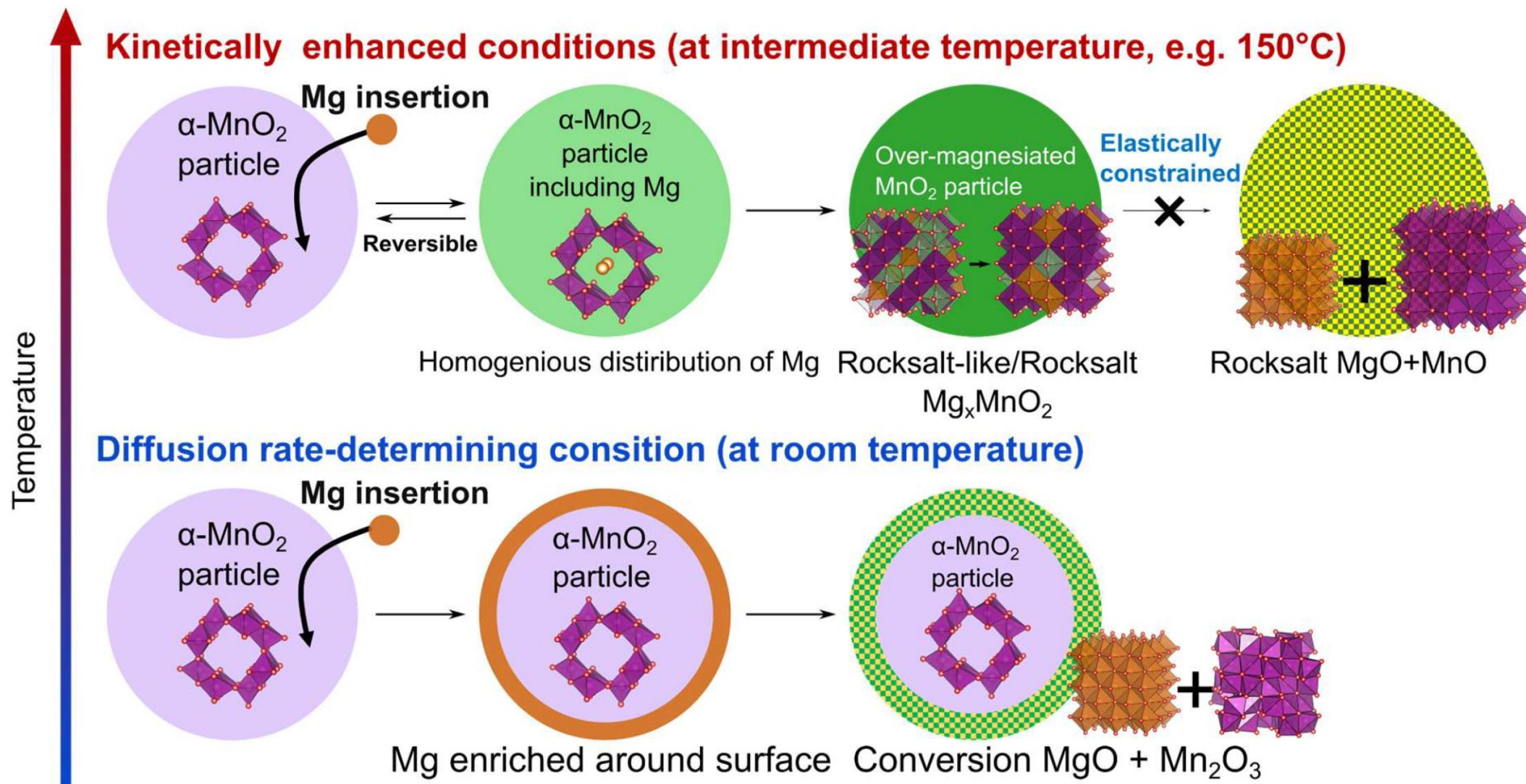
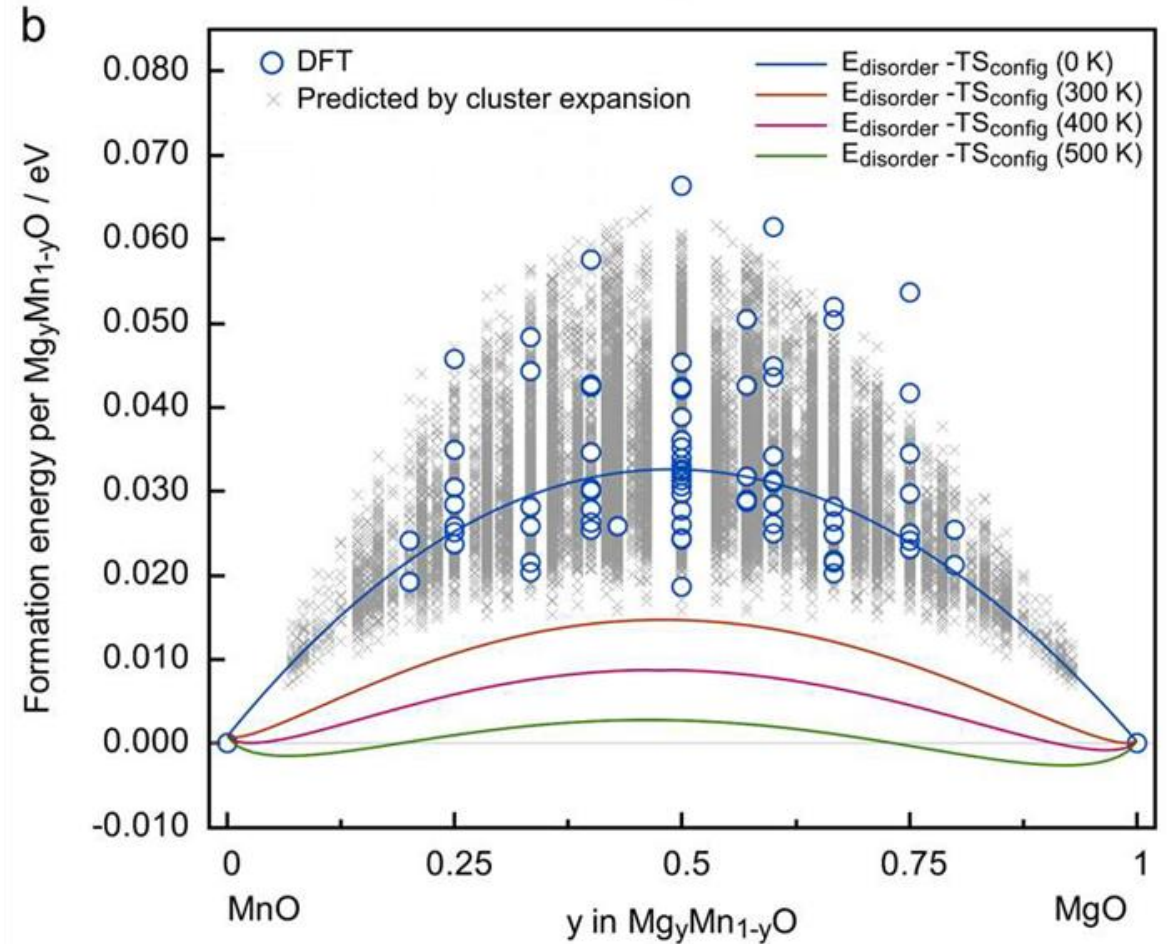
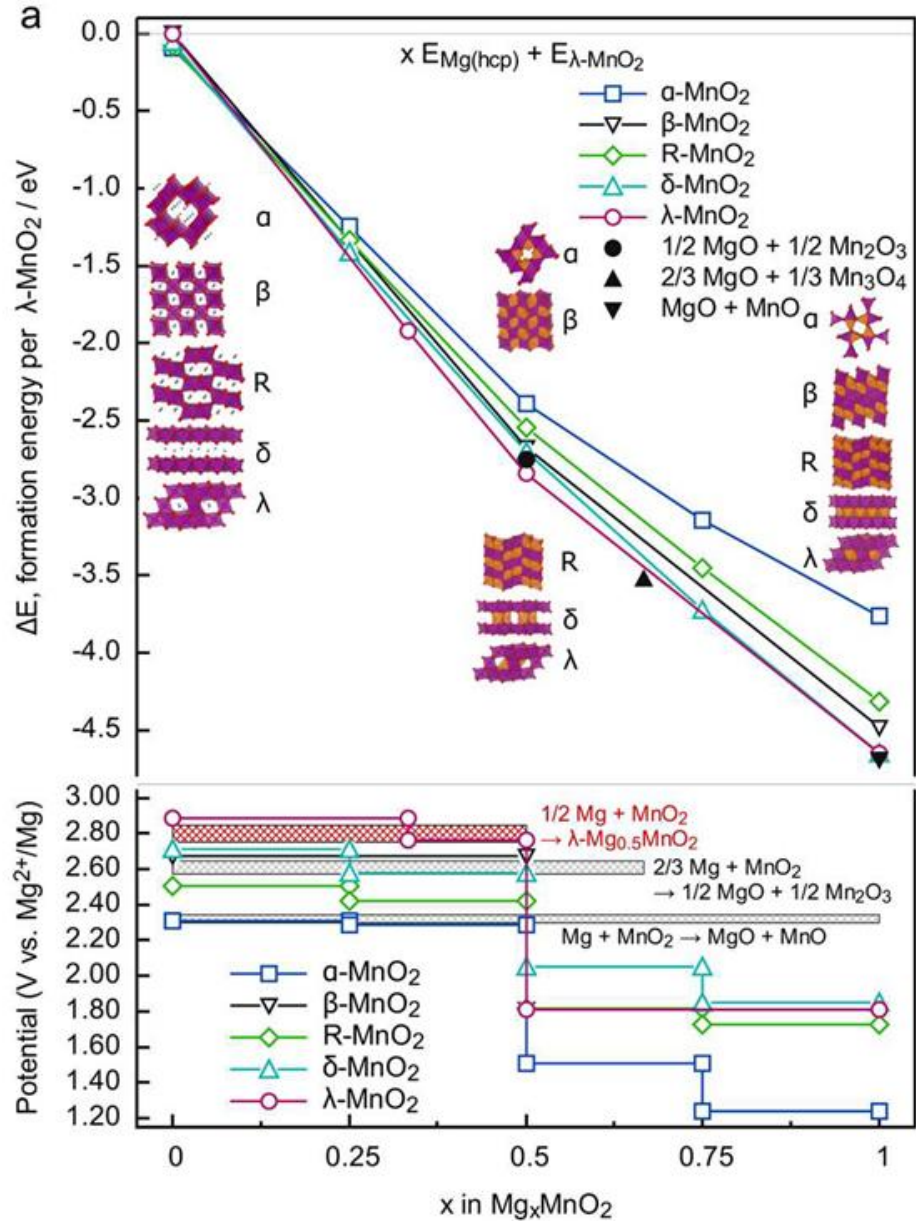


Figure 8. Schematic illustration showing the two kinds of magnesiation pathways of α - MnO_2 particles at intermediate temperature and room temperature, where the latter pathway was based on the previous work by Arthur et al.²⁵

第一原理計算によるエネルギーと電位



Accelerated Kinetics Revealing Metastable Pathways of Magnesiatio*n*-Induced Transformations in MnO₂ Polymorphs

Takuya Hatakeyama,^{*} Hongyi Li, Norihiko L. Okamoto, Kohei Shimokawa, Tomoya Kawaguchi, Hiroshi Tanimura, Susumu Imashuku, Maximilian Fichtner, and Tetsu Ichitsubo^{*}



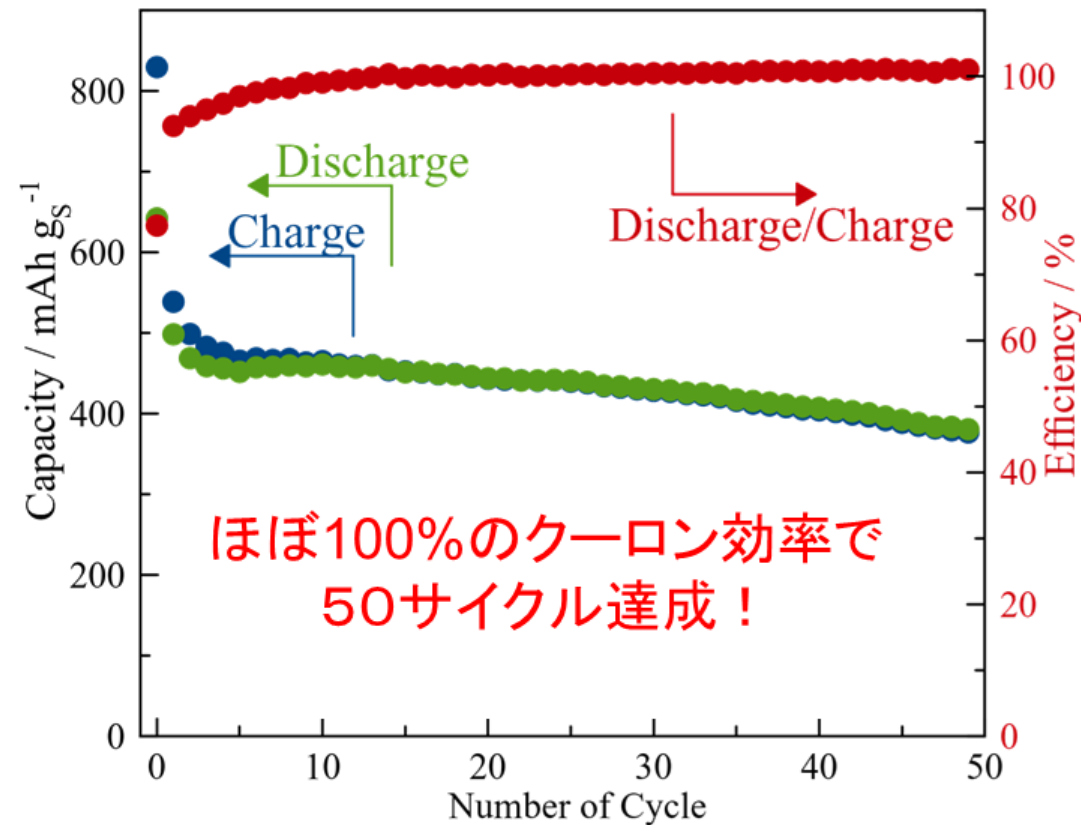
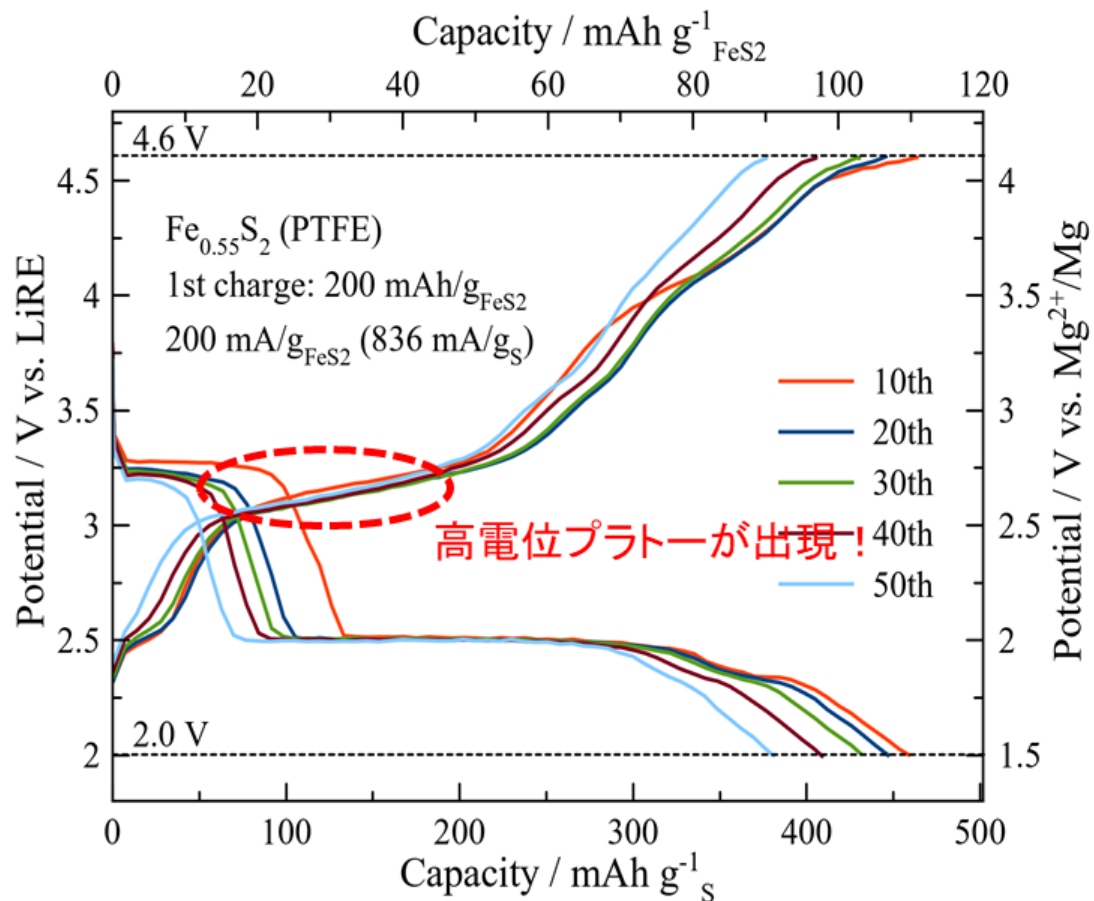
Cite This: *Chem. Mater.* 2021, 33, 6983–6996



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液体S/硫化物複合正極

バインダーをPVdFからPTFEに変更 (FeS_2 : Carbon : PTFE = 70 : 25 : 5)



高電位プラトーを利用した可逆的なサイクルを実現

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Cite this: *J. Mater. Chem. A*, 2021, 9, 16585

Electrochemically synthesized liquid-sulfur/sulfide composite materials for high-rate magnesium battery cathodes†

Kohei Shimokawa,^{id}*^{ab} Takuya Furuhashi,^a Tomoya Kawaguchi,^{id}^a
Won-Young Park,^a Takeshi Wada,^a Hajime Matsumoto,^{id}^c Hidemi Kato^a
and Tetsu Ichitsuho^{id}*^a

Mg/S batteries are some of the most promising rechargeable batteries owing to their high theoretical energy density. Their development is, however, hindered by (i) low electronic conductivity of S, (ii) sluggish Mg²⁺ diffusion in solid Mg–S compounds formed by discharge, and (iii) dissolubility of polysulfides into electrolytes. To address these problems, we propose liquid-S/sulfide composite cathode materials in combination with an ionic liquid electrolyte at intermediate temperatures (~150 °C). The composite structure is spontaneously fabricated by electrochemically oxidizing metal sulfides, yielding liquid S embedded in a porous metal-sulfide conductive frame. This concept is demonstrated by a S/FeS₂ composite cathode, which shows a significantly high-rate capability of, e.g., 1246 mA g⁻¹_(S) with a capacity of ~900 mA h g⁻¹_(S). In addition, non-equilibrium liquid S formed by fast charging results in an unexpected higher discharge potential. This work provides a new strategy to design S-based cathodes for achieving high-rate multivalent rechargeable batteries.

Received 25th April 2021
Accepted 28th June 2021

DOI: 10.1039/d1ta03464b

rsc.li/materials-a